Preparation and physical properties of a Cr3Al film with a DO3 structure

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Abstract
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Keywords
physical, do3structure, preparation, filmwith, cr3al, properties

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A Cr₃Al compound with a DO₃ structure has previously been predicted to be nearly half metal and a promising spintronics material; however, its synthesis has not been reported. Here, a Cr₃Al compound with a DO₃ structure is successfully prepared in thin-film form by the magnetron sputtering method. It was found that the substrate temperature is crucial to the atomic ordering, thin-film density and lattice constant. The lattice constant varies with different substrate temperatures and is smaller than the theoretical equilibrium lattice constant. Theoretical investigations on the electronic structures and magnetic properties indicate that the Cr₃Al compound with a DO₃ structure is a rare material with zero-gap half-metallic characteristics under an experimental lattice constant of 5.83 Å. The experimental result is in agreement with the theoretical results in magnetization, and the Cr₃Al compound synthesized in this work exhibits semimetallic-like electrical transport characteristics and positive magnetoresistance of greater than 2% in the temperature range 2–250 K.

1. Introduction

Materials that can supply highly spin polarized electrons or current are strongly desired in spintronics devices because the performance of such devices greatly depends on the spin polarization of the current or some key materials. The half-metallic materials (HMs) are generally characterized by a metallic band structure for one spin channel and a semiconductor/insulative band structure for the other spin channel. Such distinctive electronic band structures of HMs leads to a 100% spin polarization of electrons at the Fermi level ($E_f$) and makes HMs one the most promising spintronics materials (Katsnel’son & Irkhin, 1994; Zútíć et al., 2004; Dieny et al., 1991; Schmidt et al., 2000; Fiederling et al., 1999; DeGroot et al., 1983). In recent years, a type of special HM [also called spin-gapless semiconductors (SGS)] has attracted more attention. For SGS, there is a zero gap at the Fermi level in the majority spin channel and the mobility of carriers is stronger than that in common semiconductors and not only the electrons but also the holes can be fully spin-polarized (Wang, 2008; Wang et al., 2016, 2010; Zhang et al., 2015; Xu et al., 2013).

The concept of SGS was first proposed in the diluted magnetic Co-doped PbPdO₂ by Wang et al. (2008). In 2013, Gao and Yao theoretically designed several high-spin-polarization materials in binary DO₃-type structural $X_3Z$ ($X = Sc, Ti, V, Cr, Mn, Fe; Z = Al, Si, Ga, Ge$) compounds. Their
Calculation results show that Cr$_3$Al compounds have a high spin polarization at the Fermi level in a wide range of lattice constant and can be an SGS when the lattice constant is up to 6.22 Å (Gao & Yao, 2013). The Cr$_3$Al compound with a DO$_3$ structure is one of the few binary alloys with high polarization and SGS characteristics reported in previous theoretical studies. The experiments on the synthesis and physical properties of the Cr$_3$Al film were carried out in 2012 by Boekelheide and coworkers. They used electron-beam evaporation to prepare Cr$_3$Al thin films, but no samples with a perfect DO$_3$ structure were successfully synthesized. Boekelheide pointed out that the X phase (in fact, the X phase is just a DO$_3$ structure with an inhomogeneous distribution of Al atoms) is the most stable and was synthesized as a film by the electron beam evaporation method using a Cr$_3$Al alloy. At the same time, the magnetic moment of the X phase was found to be far lower than that of the DO$_3$ structure for the Cr$_3$Al compound, which is consistent with theoretical predictions (Boekelheide et al., 2012a, b, 2010).

The unsuccessful experimental synthesis, the diverse band structure and physical properties of the Cr$_3$Al compound with a DO$_3$ structure predicted in previous work motivate us to carry out further experimental and theoretical investigations. In this paper, we attempted to prepare a Cr$_3$Al film using the magnetron sputtering method. The Cr$_3$Al compound with a DO$_3$ structure has been successfully synthesized in thin-film form. We will show that a practical lattice constant of 5.83 Å is smaller than the theoretical equilibrium lattice constant. The electronic structure will also be calculated and discussed for the Cr$_3$Al compound with such a smaller lattice constant is a material with the rare zero-gap half-metallic characteristics. The experimental result is in agreement with the theoretical result in magnetization. The semi-metallic-like electrical transport characteristics and large positive magnetoresistance are observed and discussed for the Cr$_3$Al compound with a DO$_3$ structure in this paper.

2. Experimental and computational details

Cr$_3$Al films were prepared by an adjustable three-target ultra-high vacuum magnetron sputtering system with a base pressure below $3 \times 10^{-5}$ Pa. The Cr and Al targets used to prepare Cr$_3$Al compound films are made from pure metals with a purity higher than 99.9%. The substrates used in this work are 25 × 14 mm glass slides. Cr$_3$Al compound films were deposited from two-target co-sputtering. The Ar pressure during sputtering was 1.0 mTorr and the deposition rate was about 0.5 nm s$^{-1}$. The substrate temperature is selected at 50–400°C and the sputtering time is 30 min. In the rest of the paper, we will use ‘sample 100, sample 150…’ to denote the samples prepared at substrate temperatures of 100, 150°C etc. The structural properties of the sample were investigated by X-ray diffraction (XRD) using Cu Ka radiation. The magnetic properties were detected by a vibrating sample magnetometer. The M–H curve of the sample with substrate was measured first and then the film was peeled off the substrate and the M–H curve of the substrate was subsequently measured. Finally, the M–H curve of the pure film was achieved by deducting the data of the substrate from the M–H curve of the sample with substrate. Surface morphology was observed using a scanning electron microscope (SEM). SEM was also used for energy-dispersive X-ray spectroscopy (EDS) to ensure the chemical composition of the film had not deviated from the Cr$_3$Al stoichiometry. We used two methods to determine the density of the unit cell. One is in terms of the lattice constant achieved from the X-ray diffraction patterns. The other is according to the practical measurement of the mass and volume. The mass was measured by an electronic balance with an accuracy of 0.00001 g, which gave an error of less than 1% for the mass measurement of our film samples. The film thickness measurement of the sample was performed using a Dektak 6M-Stylus Profiler film thickness tester. The electronic transport measurements were performed using a physical property measurement system.

The electronic structure calculations were performed using the WIEN2K package based on the full-potential linearized

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**Figure 1**

SEM images of (a) sample 100, (b) sample 150 and (c) sample 200.
augmented plane-wave method (Blaha et al., 1990; Saini et al., 2013; Elk FP-LAPW code, version 1.4.22 http://elk.sourceforge.net). The generalized gradient approximation in the Perdew–Bueke–Ernzerhof scheme was used to implement electronic-exchange related functions (Hafner et al., 2002; Graf et al., 2011; Hsu et al., 2002; Perdew et al., 1996). The muffin-tin radius used in the calculations is generated by the system. A converged ground state was obtained using 10000 \(k\) points in the first Brillouin zone with \(K_{\text{max}} R_{\text{MT}} = 8.0\) (\(K_{\text{max}}\) represents the maximum size of the reciprocal-lattice vectors and \(R_{\text{MT}}\) is the muffin-tin radius). Wavefunctions and potentials inside the atomic sphere are expanded in spherical harmonics up to \(l = 10\) and 4, respectively.

3. Results and discussion

3.1. Morphology and structure characterization of the Cr\(_3\)Al film

Surface morphology of samples 100, 150 and 200 were observed by SEM and the images obtained of the sample surface are shown in Fig. 1. It is very clear that all the samples show the grain structure. From the perspective of overlooking the film surface, a single grain is in a triangular shape with sharp edges and a size of about 150 nm. The size of the grains are essentially uniform in the films. Thickness measurements show that the grain distribution and the film thickness are homogeneous in several square centimetres. The size of grains is similar and the space between grains has no obvious difference for samples 100, 150 and 200. The EDS shows the composition of the grains to be Cr\(_3\)Al, with an uncertainty of \(\pm 3\%\) with respect to each element, i.e. very close to the intended stoichiometry for all the used sample in this paper. Detailed results on the film composition are shown in Table 1.

The DO\(_3\) structure has an \(Fm\overline{3}m\) space group (No. 225) and can generally be seen as a set of four interpenetrating face-centered-cubic (f.c.c.) lattices, with A (0, 0, 0) B (1/4, 1/4, 1/4) C (1/2, 1/2, 1/2) and D (3/4, 3/4, 3/4) arranged along the space diagonal in the Wyckoff coordinates. For the Cr\(_3\)Al compound with a DO\(_3\) structure, the Cr atoms occupy A, B and C sites and Al atoms occupy D sites as illustrated in the inset of Fig. 2. The order-independent principal peak reflections (220), (400) and (422) can be observed in the standard XRD powder diffraction patterns for a material with a perfect DO\(_3\) structure and the (111) and (200) peaks that correspond to the order-dependent superlattice reflections are much weaker in intensity than (400) in intensity. As a result, they cannot be observed in XRD patterns. Even so, our XRD patterns can still show that all the samples crystallize in a pure phase with a DO\(_3\)-structure framework, although the lack of (111) and (200) diffraction peaks lead to an uncertainty in the ordering degree of Cr–Al. Comparing the intensities of the peak of (400) is not observed due to its own very weak intensity and the effects of texture and peak widening originating from the thin-film form. The (111) and (200) peaks correspond to the order-dependent superlattice reflections and are several times weaker than (400) in intensity. Figure 2 shows the XRD patterns of samples 100, 150 and 200. The XRD patterns shown here are quite similar to the results reported by Boekelheide et al. (2012a) where the film samples were prepared by electron beam evaporation. The structure determined using XRD patterns is f.c.c. based and the order-independent principal peak reflections of (220) and (422) can be clearly observed for all samples prepared at different substrate temperatures. Also, no peaks of impure phases were observed in any of the XRD patterns. The order-independent principal

### Table 1

<table>
<thead>
<tr>
<th>Film thickness (A)</th>
<th>Density ((\times 10^3) kg m(^{-3}))</th>
<th>Atomic ratio</th>
<th>(a) (A)</th>
<th>(M_t) ((\mu_\text{B}))</th>
</tr>
</thead>
<tbody>
<tr>
<td>Sample 100</td>
<td>8494.2</td>
<td>6.08</td>
<td>3.00:1</td>
<td>5.83</td>
</tr>
<tr>
<td>Sample 150</td>
<td>9518.3</td>
<td>5.72</td>
<td>2.98:1</td>
<td>5.88</td>
</tr>
<tr>
<td>Sample 200</td>
<td>10088.6</td>
<td>5.29</td>
<td>2.99:1</td>
<td>5.91</td>
</tr>
</tbody>
</table>

Figure 2

Theoretical powder XRD pattern for the Cr\(_3\)Al compound with a DO\(_3\) structure; the inset is the structure model.

Figure 3

X-ray diffraction patterns of samples 100, 150 and 200.
(220) peak with the (422) peak, one can see that the crystal grains of sample 100 have a greater preference for the (220) orientation than those of sample 200. So, it is possible that sample 100 has no (400) diffraction peak in the XRD pattern, instead, sample 200 has this peak. As shown in previous work, the magnetization and magnetic structure are quite sensitive to the ordering degree of Cr–Al (Boekelheide et al., 2012a, b, 2010). So, with the help of the magnetic measurements, we can determine the atomic ordering degree in the Cr3Al film, which will be investigated and discussed in detail in Section 3.2.

The lattice constants determined here are 5.83, 5.88 and 5.91 Å, and the film thicknesses are 8542.5, 9598.6 and 10068.6 Å for samples 100, 150 and 200, respectively. This indicates that the lattice constant increases and the Cr3Al compound density increases with increasing substrate temperature. Furthermore, when the substrate temperature is higher than 200°C, the lattice constant is stable at 5.91 Å, and the film thickness is also almost unchanged with changing substrate temperature. The data on the lattice constant and film thickness are also collected in Table 1. All the lattice constants are smaller than the theoretically predicted equilibrium lattice constant of 5.92 Å (Gao & Yao, 2013). In the work by Boekelheide et al. (2012a), the lattice constant is reported to be 5.90 Å for the Cr3Al compound film prepared by electron-beam evaporation (sample EBE), which is fairly consistent with the results determined in sample 200 or the samples prepared at substrate temperatures higher than 200°C. In fact, the magnetization is also very close for the reported sample EBE and our sample 200, which will be shown and discussed in the next section.

3.2. Magnetic properties and electronic structure

Fig. 4 shows the magnetization curves measured at 2 K for samples 100, 150 and 200. The total magnetic moments of the unit cell (Mt) achieved from the magnetization curves are shown in Table 1. Mt is 1.1 μB for sample 200, and was reported to be 1.06 ± 0.02 μB for sample EBE in Boekelheide et al. (2012a). It is clear that Mt is comparable for sample 200 and sample EBE. It is reported that sample EBE does not crystallize in a perfect DO3 structure but in an X-phase structure. The inhomogeneous distribution of Al atoms in Cr3Al causes the antiferromagnetic arrangement of the Cr atomic magnetic moment and a decrease in magnetization (Boekelheide et al., 2012a, b, 2010). According to first-principles calculations, Cr3Al with a perfect DO3 structure should have an Mt of 2.92–3 μB when the lattice changes in the range 5.90–6.22 Å (Gao & Yao, 2013). To associate this with the lattice constant mentioned in Section 3.1 would imply that sample 200 has the same atomic occupation and ordering as sample EBE.

Furthermore, from Fig. 4 it can be seen that the saturation magnetization (namely, Mt) increases with decreasing substrate temperature. When the substrate temperature reaches 100°C, the samples show an Mt of about 2.88 μB. The Mt of 2.88 μB is slightly smaller that the theoretical results reported by Gao & Yao (2013), where the Cr3Al compound is hypothetically in a perfect DO3 structure with a lattice constant of 5.90–6.22 Å. In fact, it should be noted that the lattice constant contracts with decreasing substrate temperature. For sample 100, the lattice constant is 5.83 Å, which is smaller than that used in the work by Gao & Yao (2013). In order to compare the experimental and theoretical results more clearly and accurately, the electronic structure and magnetic properties were calculated by first-principles calculations for the Cr3Al compound with a perfect DO3 structure and a lattice constant of 5.83 Å. The calculated atomic
magnetic moments are 1.98 $\mu_B$, $-1.29 \mu_B$ and $0.03 \mu_B$ for Cr (A, C), Cr (B) and Al atoms, and the $M_t$ is 2.88 $\mu_B$ which is quite consistent with our experimental result of 2.88 $\mu_B$. This indicates that the Cr$_3$Al compound with a perfect DO$_3$ structure was successfully synthesized as a film at the substrate at 100°C by the magnetron sputtering method. In addition, Fig. 4(a) shows an enlarged part of the M–H curves at low field. One can see that the coercivity of Cr$_3$Al is very small and Cr$_3$Al has a fairly soft magnetic characteristic. Fig. 4(b) shows the temperature dependence of magnetization (M–T curve) in the field of 500 Oe. It is clear from the M–T curve that the Curie temperature of Cr$_3$Al is 250 K.

The calculated band structures for the Cr$_3$Al compound with a lattice constant of 5.83 Å are shown in Fig. 5. For comparison, we also provide the band structures at the equilibrium lattice parameter and at a lattice parameter of 6.22 Å. It is clear that these band structures are very similar. However, it can also be seen that at the lattice constant of 5.83 Å, the Cr$_3$Al compound with a perfect DO$_3$ structure is no longer a common half-metal or spin-gapless semiconductor for the Fermi level across the valence band top and the vanishing of the band gap in the spin-up channel. More importantly, such a band structure is characteristic of a zero-gap HM; hence, the Cr$_3$Al compound synthesized in this work can be considered to be a zero-gap HM. The concept of a zero-gap HM was first proposed by Du et al. (2013). So far, the zero-gap half-metal as a special spintronics material has rarely been reported. However, many novel physical properties occurring in the zero-gap HM – for example, the crossover of magnetoresistance – have considerable theoretical research value and potential applications in the field of spintronics.

3.3. Electrical transport properties

Fig. 6 shows the dependence curves of resistance on temperature with and without the magnetic field for sample 100. It can be seen that the resistance of the Cr$_3$Al compound decreases with increasing temperature and exhibits a semimetallic-like behavior. The inflection points occurring near 250 K correspond to the Curie temperature achieved from the M–T curve. The total DOS and spin-resolved DOS patterns are shown in Fig. 7. It can be observed that the Fermi level always lies in a deep valley for spin-up, spin-down or the total DOS. Usually this deep valley is considered as a pseudo-gap and the semi-metallic-like electrical transport behavior can be attributed to the pseudogap-type electronic structure for Cr$_3$Al compounds. Below the Curie temperature, the resistance with the magnetic field of 5 T is obviously higher than that without a magnetic field, which indicates a positive magnetoresistance behavior and the related data are plotted in Fig. 6(b). The positive magnetoresistance is higher than 2% and the maximum of about 5% occurs near the Curie temperature. Usually a positive MR can be widely observed in nonmagnetic systems. However, the special cases are the zero-gap half-metals and the spin-gapless semiconductors (Du et al., 2013; Ouardi et al., 2013). Du et al. (2013) and Ouardi et al. (2013) reported the positive MR of Fe$_2$CoSi and Mn$_2$CoAl Heusler compounds at low temperature and attributed this characteristic to the unique band structures, that is, spin-gapless semiconductive or zero-gap half-metallic band structures. As illustrated on the basis of band structures of Cr$_3$Al shown in Fig. 5, the Cr$_3$Al compound with the experimental lattice constant has a zero-gap half-metallic band structure. Therefore, the large positive magnetoresistance can be thought to be closely related to the zero-gap half-metallic characteristics of the band structure for the Cr$_3$Al compound.

4. Conclusions

A Cr$_3$Al compound with a DO$_3$ structure has been successfully synthesized in the film form by the magnetron sputtering
method. An experimental lattice constant of 5.83 Å is achieved and is smaller than the theoretical equilibrium lattice constant. According to first-principles calculations, it was found that the Cr₃Al compound with a DO₃ structure is a material with rare zero-gap half-metallic characteristics in a much smaller lattice constant. The experimental result is in agreement with the theoretical result in magnetization. The semi-metallic-like electrical transport characteristics and large positive magnetoresistance are considered to originate from the special electronic structures of the Cr₃Al compound with a smaller lattice constant.

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