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Enhanced materials performance of Al/Ti/Al laminate sheets subjected to cryogenic roll bonding

Abstract

Laminate sheets attract increasing attention from researchers and engineers. In this paper, Al/Ti/Al laminate sheets were fabricated by using cryogenic roll bonding for first time. The edge defects, mechanical properties, and interface bonding of laminate sheets by cryogenic roll bonding technique were compared with these by room-temperature roll bonding technique. Results show that there are some edge cracks in laminate sheets by room-temperature roll bonding while they do not appear when subjected to cryogenic roll bonding. The ultimate tensile stress of laminate sheets by cryogenic roll bonding increases up to 36.7% compared to that by room-temperature roll bonding. When laminate sheets are rolled to 0.125 mm from 2.025 mm, the interfaces between Al and Ti layers are bonded well for both cryogenic roll bonding and room-temperature roll bonding. Finally, we discussed the improvement in edge quality and mechanical properties and the mechanism of interface bonding of Al/Ti/Al laminate sheets during cryogenic roll bonding.

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Enhanced materials performance of Al/Ti/Al laminate sheets subjected to cryogenic roll bonding

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Abstract: Laminate sheets attract increasing attention from researchers and engineers. In this paper, Al/Ti/Al laminate sheets were fabricated by using cryogenic roll bonding for first time. The edge defects, mechanical properties and interface bonding of laminate sheets by cryogenic roll bonding technique were compared with these by room-temperature roll bonding technique. Results show that there are some edge cracks in laminate sheets by room-temperature roll bonding while they do not appear subjected to cryogenic roll bonding. The ultimate tensile stress of laminate sheets by cryogenic

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roll bonding increases 36.7% compared to that by room-temperature roll bonding. When laminate sheets are rolled to 0.125 mm from 2.025 mm, the interfaces between Al and Ti layers are bonded well for both cryogenic roll bonding and room-temperature roll bonding. Finally, we discussed the improvement in edge quality and mechanical properties and the mechanism of interface bonding of Al/Ti/Al laminate sheets during cryogenic roll bonding.

Key words: Cryorolling; roll bonding; Al/Ti/Al laminate; edge defect; mechanical property; interface bonding

I. INTRODUCTION

In recent years, ‘sandwich’ laminate sheets have attracted a great deal of attention due to their excellent properties [1-4] which make them ideally suited for diverse functions such as ballistic protection, heat exchange, blast mitigation and vibration damping [5-6]. Roll-bonding techniques have been successfully applied to fabricate multi-component materials with tailored properties by reinforcement or grading, such as Al/Ti [7-12], Cu/Ti [13, 14], Al/Mg [15, 16], Al/Cu [17, 18] and Al/Steel [19, 20].

For laminate sheets, rolling temperature generally affects the interface bonding quality and mechanical properties of the product [21-23]. It has been observed that in order to obtain good interface bonding quality, it is better to raise the deformation temperature [24]. For ferrous metals, rolling at high temperature ($>1273\text{K}$) is required to achieve good interface bonding [23, 24]. Interfaces of non-ferrous metals can be bonded well under moderately high temperature rolling ($\sim 473\text{K}$) [25, 26]. The bonding

strength of metals at room-temperature roll bonding ($\sim 293\text{K}$) is related to the reduction ratio [27]. It is seen that when the reduction ratio is larger than 70%, the metals can bond well [28, 29]. Compared to symmetric roll bonding, the bonding quality between layers can also be improved by asymmetric roll bonding [30, 31]. However, there are no reports on the fabrication of laminate sheets by cryogenic roll bonding technique.

Cryorolling ($<173\text{K}$), in which the samples are generally cooled by liquid nitrogen, has been considered as a severe plastic deformation technique that can improve both the strength and the ductility of ultrafine-grained materials [32-26]. Recently, Yu et al [37] found that the cryorolling can suppress the dynamic softening of UFG Al sheets subjected to accumulative roll bonding (ARB) at room temperature. Sheets subjected to (ARB + cryorolling) show improved mechanical properties compared to ARB-processed sheets. However, to the best of the authors' knowledge, there have been no reports on the mechanism of interface bonding of laminate sheets subjected to cryogenic roll bonding.

There are some reports that Al/Ti/Al laminate sheets fabricated by room-temperature roll bonding [7-12]. Yu et al [7] fabricated a trimodal laminate sheets combined with coarse-grained Al, ultrafine-grained Ti and TiAl_3 particles combined room-temperature roll bonding and annealing. Luo and Acoff [12] fabricated multilayered Ti/Al laminate sheets by room-temperature roll bonding and annealing. They found that TiAl_3 layer forms at low temperature in the Al-rich layer after just a few hours at a temperature of 923 K. In this study, the Al/Ti/Al laminate sheets were fabricated by using cryogenic roll bonding for first time. The quality of laminate sheets by cryogenic roll bonding

such as edge cracks, mechanical properties, and interface bonding were compared with these by room-temperature roll bonding. Finally, we discussed the improvement in edge quality and mechanical properties as well as the mechanism of interface bonding of Al/Ti/Al laminate sheets during cryogenic roll bonding.

II. EXPERIMENTAL

In this study, Al/Ti/Al laminate sheets were produced by cryogenic roll bonding and room-temperature roll bonding separately. Commercial pure aluminum alloy 1060 (AA1060) and commercial pure titanium sheets were used. Before rolling, the thicknesses of the AA1060 sheets and Ti sheets were 1.0 mm and 25 μm respectively, and the surfaces of sheets were cleaned by alcohol. The laminate sheets were stacked as Al/Ti/Al laminate sheets. Before rolling, the front end of laminate was welded by spot welding to prevent slipping. Then they were processed by both room-temperature roll bonding and cryogenic roll bonding separately. A 4-high multifunction rolling mill with 50 mm diameter work rolls was used. The thicknesses of laminate sheets were 0.9 mm, 0.5 mm, 0.25 mm and 0.125 mm after the first, second, third and fourth rolling pass respectively. Fig. 1 shows an illustration of laminate subjected to the cryogenic roll bonding technique. During cryogenic roll bonding, the laminate sheets were cooled by liquid nitrogen for at least 8 min before the rolling commenced.

FIG. 1. Illustration of Al/Ti/Al laminate sheets during cryogenic roll bonding.

After rolling, the edge quality of laminate sheets was analyzed. The tensile tests were carried out for both kinds of laminate sheets with tensile test samples with 25 mm gauge length and 6 mm gauge width with a strain rate of $1.0 \times 10^{-3} \text{ s}^{-1}$ on an INSTRON machine to determine the engineering strain vs engineering stress characteristics. In order to study the interface bonding behavior, we used the focused ion beam technique to prepare transmission electron microscope (TEM) film samples on an FEI xT Nova Nanolab 200 Dual-beam workstation. The details of the interface between the Ti and Al layers were analyzed by a Philips CM200 field emission gun TEM operating at 200 kV.

III. RESULTS

Figs. 2(a1)-(c1) show the Al/Ti/Al laminate sheets produced by room-temperature roll bonding. In the figure, there are some edge cracks which occur in the laminate sheets after rolling. However, there are no edge cracks in the laminate sheets produced by cryogenic roll bonding, as shown in Figs. 2(a2)-(c2). It is obvious that the cryogenic roll bonding can enhance the quality of laminate edge compared to that by room-temperature roll bonding.

FIG. 2. (a1, b1 and c1) for room-temperature roll bonded sheets, (a2, b2, and c2) for cryogenic roll bonded sheets. (a1), (a2) the second pass; (b1), (b2) the third pass; (c1), (c2) the fourth pass.

Fig. 3 shows the engineering strain vs engineering stress curves of Al/Ti/Al laminate sheets after the fourth pass subjected to room-temperature roll bonding and cryogenic roll bonding respectively. For the laminate sheets subject to room temperature rolling, the ultimate tensile stress is 150MPa, which reaches 205 MPa for the laminate sheet subject to cryogenic roll bonding. The ultimate tensile stress increases 36.7% by cryogenic roll bonding compared to that by room-temperature roll bonding.

FIG. 3. Engineering strain vs stress of Al/Ti/Al laminate sheets by cryogenic roll bonding and room-temperature roll bonding.

Figs. 4(a) and 4(b) show the TEM images of the interface between the Ti and Al layers after the second and the fourth pass of room-temperature roll bonding. After the second pass, some small voids are visible at the interfaces, although over most of the interface, the Al and Ti layers bond well. After the fourth pass, the interface between Ti and Al layer shows good bonding. Figs. 4(c) and 4(d) show TEM images of the interface between the Ti and Al layers after the second and the fourth passes of cryogenic roll bonding. Compared to the samples produced by room-temperature roll bonding, the residual voids are much larger at the interface and the grain sizes of Ti and Al layers near the interface are much finer after the second pass of cryogenic roll bonding. This is relevant to the finding that the bonding between interfaces becomes difficult with reduced temperature [27]. With a higher number of rolling passes, the grains near the interface grow during rolling, and then the interface between Al and Ti layers also

shows a good bond, as seen in Fig. 4(d).

FIG. 4. TEM images of the microstructure near interface between Al and Ti layers. (a) and (b) for room-temperature roll bonding after the second and fourth passes, (c) and (d) for cryogenic roll bonding after the second and fourth passes.

IV. DISCUSSION

1) EDGE CRACKS

Edge cracks are the main defects in laminate products [38-41]. These edge cracks must be trimmed before subsequent processing. For some metals such as Mg and steel, the ductility decreases at lower temperature. However, the ductility of Al alloys at cryogenic temperature is higher than that at room temperature. Glazer et al [42] studied the tensile curves of AA2090 at 298 K, 77K and 4K, and found that the ductility of this Al alloy increases at reduced temperature. Park et al [43] also studied the stress-strain curves of AA5052, AA5086, AA6061 and AA6082 sheets at temperatures ranging from 293 K to 110K. They found that the fracture strain increases with reduction in deformation temperature.

Fig. 5(a) and Fig. 5(b) show the TEM images of the microstructure of Al layers after the fourth pass subjected to cryogenic roll bonding and room-temperature roll bonding respectively. In Fig. 5(a), there are no voids in the sample. However, there are a number of nanoscale voids in the sample subjected to room-temperature roll bonding, as shown

in Fig. 5(b). This means that the Al layer will have higher ductility during cryogenic roll bonding compared to that during room-temperature roll bonding. Thus it suggests that a much lower deformation temperature can potentially be applied to reduce the edge-cracking tendency of Al laminate sheets, as shown in Fig. 2.

FIG. 5. TEM image of microstructure of Al layers after the fourth pass of (a) cryogenic roll bonding and (b) room-temperature roll bonding. Arrows show nanoscale voids.

2) MECHANICAL PROPERTIES

Cryorolling has been reported that it can improve the mechanical properties of single metal sheets [44, 45]. Zhang et al [46] found that the ultimate tensile stress of cryorolling + ageing processed Cu-1Cr-0.1Zr alloy strips increases to 690.1MPa compared to that of 478.2 MPa by room-temperature rolling, which increases 44.3%. For the alloy fabricated using cryorolling + ageing processing, the twin/matrix lamellar thickness reduces 55% compared to that fabricated using room-temperature rolling [46]. Naga Krishna et al [47] found that the mechanical properties of cryorolled Al-Mg-Si alloy improved greatly compared to that of room-temperature rolled samples. The enhanced properties of cryorolled sheets are due to the high dislocation density, increased microstrain and a reduced crystallite size compared to room-temperature rolled sheets [47]. In this study, we also believe that the finer grain size by cryogenic

roll bonding improves the mechanical properties of laminate sheets compared to that by room-temperature roll bonding. In Fig. 5, we can find the grain size of aluminium layer subjected to cryogenic roll bonding is much finer than that subjected to room-temperature roll bonding. Fig. 6(a) and Fig. 6(b) show the TEM images of microstructure of Ti layer after the fourth pass subjected to cryogenic roll bonding and room-temperature roll bonding respectively. In Fig. 6, it is obvious that the grain size of Ti layer subject to cryogenic roll bonding is much finer than that subjected to room-temperature roll bonding. According to Hall-Petch formula, the yield strength of metal materials improves with finer grain size. Thus the strength of Al/Ti/Al laminate sheets by cryogenic roll bonding is much higher than that by room-temperature roll bonding, as shown in Fig. 3.

FIG. 6. TEM image of microstructure of Ti layers after the fourth pass of (a) cryogenic roll bonding and (b) room-temperature roll bonding.

3) INTERFACE BONDING

Interface bonding is one of the important factors that determine the final quality of product. There are many factors that determine the bonding quality such as rolling temperature, reduction ratio [27]. In this study, the rolling reduction ratio per pass by cryogenic roll bonding is the same as that by room-temperature roll bonding. Thus, the rolling temperature results in the difference of interface bonding behavior as shown in

Fig. 4.

Fig. 7 shows the distribution of grain size of Al layer and Ti layer after different processing. As shown in Fig. 7(a), the grain size of Al near the Al/Ti interface is 186 nm for samples rolled at room temperature, while the grain size is refined to 48 nm for the cryorolled samples as shown in Fig. 7(b). In addition, the grains in the Al layer near the interface grow significantly under the action of stress. As shown in Fig. 7(c) and Fig. 7(d), the grain size increases to 398 nm and 386 nm after the fourth pass of room-temperature roll bonding and cryogenic roll bonding respectively. It is obvious that the grain growth affects the bonding between the Al and Ti interfaces.

FIG. 7. Grain size distribution in Al layer. (a) and (b) for room-temperature roll bonding after the second and fourth pass; (c) and (d) for cryogenic roll bonding after the second and fourth pass.

To date, some mechanisms for room-temperature roll bonding have been presented, such as the film theory, energy barrier theory, and diffusion bonding theory, which are reviewed in Refs. [21, 27]. Here, we propose a new bonding mechanism based on grain refinement and grain growth of soft layer. Fig. 8 shows the mechanisms of interface bonding between Al and Ti layer subjected to room-temperature roll bonding and cryogenic roll bonding. Before rolling, the Al and Ti surfaces are separate, and the grains are coarse, as shown in Fig. 8(a). During rolling, firstly, the surfaces come into contact and the grains near the interface are refined greatly. There is a great deal of

sliding between the Al and Ti sheets before the bond begins to form. The difference in mechanical properties results in a severe shear deformation in the surface layers, and then the grains are refined greatly. Due to the suppression of the dynamic recovery behavior at cryogenic temperature, compared with room-temperature roll bonding, cryorolling is more conducive to grain refinement [32-37], as illustrated in Fig. 8(b) and (d). In addition, during sliding, the underlying regions of fresh metal are exposed, forming additional bonds. Rolling with greater reduction in thickness, results in the growth of ultrafine grains / nano grains (UFG/NG). This facilitates the interfacial bond, as shown in Fig. 8(c) and (e). With further reduction in thickness, most regions of the interfaces bond and there is no sliding between the layers. The UFG/NGs near the interface grow under shear stress. There are many reports that UFG/NG grow during room-temperature deformation [32, 48, 49]. Liao et al [49] reported that the grains grow from 30 ± 7.5 nm to 129 ± 43 nm in electro-deposited NG Ni during high-pressure torsion. For ARB-processed Al sheets, abnormal grain growth during asymmetric rolling can result in elimination of the interface between layers [50]. Here, it is seen that the grains grow in the soft layer under severe shear stress both at room temperature and cryogenic temperature, which also contributes to the interfacial bonding.

FIG. 8. Illustration of the interface bonding during rolling. (a) before rolling; (b) and (c) for the interface with higher reduction by room-temperature roll bonding; (d) and (e) for the interface with higher reduction by cryogenic roll bonding.

V. CONCLUSIONS

(1) Al/Ti/Al laminate sheets were fabricated by cryogenic roll bonding for first time. Edge cracks appear in the laminate sheets produced by room-temperature roll bonding, while they do not appear in the laminate sheets produced by cryogenic roll bonding. This is due to the improvement in ductility of Al alloys at reduced temperature.

(2) The ultimate tensile stress of Al/Ti/Al laminate sheets subject to cryogenic roll bonding is higher 36.7% compared to that subject to room-temperature roll bonding.

(3) At reduced temperature, the interface bonding between Al/Ti becomes difficult. When the reduction ratio is lower, there are many residual voids near the interface between Al and Ti in the laminate produced by cryogenic roll bonding. As the rolling reduction is increasing, the interface between the Al and Ti bonds well both for room-temperature roll bonding and cryogenic roll bonding. For the cryogenic roll bonded laminate, initially the surfaces come into contact and grains near the interface of the soft layer are refined to 48 nm. In the second stage, they grow dynamically and the interface bonds well.

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FIG. 1. Illustration of Al/Ti/Al laminate sheets during cryogenic roll bonding.

FIG. 2. (a1, b1 and c1) for room-temperature roll bonded sheets, (a2, b2, and c2) for cryogenic roll bonded sheets. (a1), (a2) the second pass; (b1), (b2) the third pass; (c1), (c2) the fourth pass.

FIG. 3. Engineering strain vs stress of Al/Ti/Al laminate sheets by cryogenic roll bonding and room-temperature roll bonding.

FIG. 4. TEM images of the microstructure near interface between Al and Ti layers. (a) and (b) for room-temperature roll bonding after the second and fourth passes, (c) and (d) for cryogenic roll bonding after the second and fourth passes.

FIG. 5. TEM image of microstructure of Al layers after the fourth pass of (a) cryogenic roll bonding and (b) room-temperature roll bonding. Arrows show nanoscale voids.

FIG. 6. TEM image of microstructure of Ti layers after the fourth pass of (a) cryogenic roll bonding and (b) room-temperature roll bonding.

FIG. 7. Grain size distribution in Al layer. (a) and (b) for room-temperature roll bonding after the second and fourth pass; (c) and (d) for cryogenic roll bonding after the second and fourth pass.

FIG. 8. Illustration of the interface bonding during rolling. (a) before rolling; (b) and (c) for the interface with higher reduction by room-temperature roll bonding; (d) and (e) for the interface with higher reduction by cryogenic roll bonding.