

University of Wollongong

Research Online

---

Faculty of Engineering and Information  
Sciences - Papers: Part B

Faculty of Engineering and Information  
Sciences

---

2017

## Predicting Steel Tensile Responses and Fracture Using the Phenomenological Ductile Shear Fracture Model

Kazeem K. Adewole  
*University of Ibadan*

Lip H. Teh  
*University of Wollongong, lteh@uow.edu.au*

Follow this and additional works at: <https://ro.uow.edu.au/eispapers1>



Part of the [Engineering Commons](#), and the [Science and Technology Studies Commons](#)

---

### Recommended Citation

Adewole, Kazeem K. and Teh, Lip H., "Predicting Steel Tensile Responses and Fracture Using the Phenomenological Ductile Shear Fracture Model" (2017). *Faculty of Engineering and Information Sciences - Papers: Part B*. 1425.  
<https://ro.uow.edu.au/eispapers1/1425>

Research Online is the open access institutional repository for the University of Wollongong. For further information contact the UOW Library: [research-pubs@uow.edu.au](mailto:research-pubs@uow.edu.au)

---

# Predicting Steel Tensile Responses and Fracture Using the Phenomenological Ductile Shear Fracture Model

## Abstract

In the literature, all the micromechanical fracture models used for predicting structural steel tensile responses and fracture are based on the ductile fracture mechanism that uses the Lode angle parameter to simulate shear fracture under low-stress triaxiality. Using the phenomenological shear fracture model that uses the shear stress ratio rather than the Lode angle parameter, this technical note presents the finite element predictions of the responses of S690 steel solid and perforated coupons under tension and of the fractures of TRIP (transformation-induced plasticity) 690 steel specimens under pure shear and combined shear and tension. The calibrated phenomenological shear fracture model parameters are obtained through a phenomenological curve-fitting process that does not involve costly laboratory tests. This technical note demonstrates that the phenomenological shear fracture model can accurately predict the responses and fracture of structural steels under tension, pure shear, and combined shear and tension.

## Disciplines

Engineering | Science and Technology Studies

## Publication Details

Adewole, K. K. & Teh, L. H. (2017). Predicting Steel Tensile Responses and Fracture Using the Phenomenological Ductile Shear Fracture Model. *Journal of Materials in Civil Engineering*, 29 (12), 06017019-1-06017019-6.

1     **Predicting steel tensile responses and fracture using the phenomenological ductile shear**  
2   **fracture model**

3                                     Kazeem K. Adewole<sup>1</sup> and Lip H. Teh<sup>2</sup> M.ASCE

4     **ABSTRACT**

5     In the literature, all the micromechanical fracture models employed for predicting structural steel  
6     tensile responses and fracture are based on the ductile fracture mechanism which employed the  
7     Lode angle parameter to simulate shear fracture under low stress triaxiality. Using the  
8     phenomenological shear fracture model that employs the shear stress ratio rather than the Lode  
9     angle parameter, this technical note presents the finite element predictions of the responses of  
10    S690 steel solid and perforated coupons under tension, and of the fractures of TRIP  
11    (Transformation Induced Plasticity) 690 steel specimens under pure shear and under combined  
12    shear and tension. The calibrated phenomenological shear fracture model parameters are  
13    obtained through phenomenological curve fitting process that does not involve costly laboratory  
14    tests. This technical note demonstrates that the phenomenological shear fracture model can  
15    accurately predict the responses and fracture of structural steels under tension, pure shear and  
16    combined shear and tension.

17  
18    **Keywords:** ductile shear fracture, ductile tensile fracture, phenomenological fracture model, steel  
19    fracture, void growth mechanism

20  

---

<sup>1</sup>Lecturer, Civil Engineering Department, University of Ibadan, Ibadan, Nigeria,  
E-mail: kkadewole@yahoo.com

<sup>2</sup>Associate Professor, School of Civil, Mining and Environmental Engineering,  
Univ. of Wollongong, Wollongong, NSW 2500, Australia E-mail: lteh@uow.edu.au

## 21 **Introduction**

22 Ductile materials exhibit two types of fracture: “ductile shear fracture” and “ductile tensile  
23 fracture” (Björklund and Nilsson 2014). Either mode or a combination of both may take place in  
24 structural steel (Hooputra et al. 2004, Björklund et al. 2013). Fracture initiation in the shear  
25 mode occurs under intense shear stresses resulting either from extensive slip on the activated slip  
26 planes and/or microcracking without any void nucleation, or from void nucleation in the slip  
27 bands (French and Weinrich 1979). In any case, the shear mode is less pressure-dependent than  
28 the tensile mode (Björklund et al. 2013).

29 In the tensile mode, fracture initiation is always associated with void nucleation under  
30 predominantly tensile stresses at high stress triaxiality (Björklund et al. 2013). The growth and  
31 coalescence of nucleated voids in this mode occur under varying stress conditions with varying  
32 fracture mechanisms as the ductile fracture propagates (Danas and Castañeda 2014). The initial  
33 growth and coalescence, which are associated with inter-void ligaments necking, occur under  
34 high stress triaxiality and pure hydrostatic stress. The growth and coalescence then takes place  
35 under intermediate and subsequently low stress triaxiality. Under the intermediate stress  
36 triaxiality, a transition from the predominantly tensile stresses-based void growth mechanism to  
37 the predominantly shear stresses-based void shearing mechanism occurs (Danas and Castañeda  
38 2012).

39 The Gurson-Tvergaard-Needleman model or Gurson-like models (Cockroft and Latham 1968,  
40 Wilkins et al. 1980, Gologanu et al. 1993) and the Rice and Tracey’s void growth model or its  
41 variants (Rice and Tracey 1969, Kanvinde & Deierlein 2006, Jia and Kuwamura 2013, Kiran  
42 and Khandelwal 2013) have been employed to simulate the void nucleation, growth, and  
43 coalescence fracture mechanism. However, these models generally fail to predict predominantly

44 shear fracture under low and negative stress triaxiality (Danas and Castañe 2012, Wen and  
45 Mahmoud 2015). A parameter that improves the simulation in the later stage of the ductile  
46 fracture mode under low stress triaxiality, called the Lode angle, has been used to simulate the  
47 void shearing effect (Xue 2008, Nahshon and Hutchinson 2008, Bai and Wierzbicki 2008, Danas  
48 and Castañe 2012, Cheng et al. 2015, Wen and Mahmoud 2015, Li et al. 2016). The Lode angle  
49 is used to distinguish the different shear stress states in three dimensions.

50 In any case, all the aforementioned models require a number of parameters (up to nine) that are  
51 to be calibrated from elaborate laboratory tests, with some requiring fractographic analysis.  
52 Costly laboratory tests are therefore required to obtain the calibrated parameters (Bernauer and  
53 Brocks 2002), which are not feasible for most design firms (Kiran and Khandelwal 2013).

54 An economical method for obtaining the required fracture parameters is the phenomenological  
55 curve fitting process. The phenomenological curve fitting process involves keeping some  
56 parameters constant while varying others during numerical simulations until the simulation  
57 results match the experimental results up to the displacement at fracture (Bernauer and Brocks  
58 2002). The parameters at which the simulated displacement at fracture matches the experimental  
59 displacement are the set of critical/calibrated fracture parameters (Bernauer and Brocks 2002).  
60 This phenomenological procedure can be used without the need for any extensive laboratory  
61 tests other than the standard tension coupon test.

62 Since fracture of structural steel could be by either the shear or the tensile mode, or a  
63 combination of both, it is important to employ models that are capable of capturing the two  
64 modes (Björklund et al 2013). To the best of the authors' knowledge, the prediction of tensile  
65 responses of civil engineering structural steel using the phenomenological shear fracture model

66 that simulates ductile tensile fracture under high stress triaxiality and employs the shear stress  
67 ratio rather than the Lode angle parameter to simulate ductile shear fracture under low/negative  
68 triaxiality has not been published. All the published fracture models on structural steel that  
69 predict both ductile tensile fracture and ductile shear fracture employ the Lode angle parameter  
70 for ductile shear fracture prediction.

71 The phenomenological shear damage and fracture model (Hooputra et al. 2004) presented in this  
72 technical note simulates the ductile tensile fracture and the ductile shear fracture involving void  
73 nucleation, void shearing and elongation, inter-voids shear localisation and void coalescence  
74 under intense shear stresses (low and negative stress triaxiality). It is employed in the present  
75 work to obtain the calibrated model parameters for S690 steel. The simulations of fractures of  
76 TRIP (Transformation Induced Plasticity) 690 steel specimens under pure shear and under  
77 combined shear and tension are also presented.

### 78 **Phenomenological shear damage and fracture model/criterion**

79 The phenomenological shear damage and fracture model employs the equivalent plastic strain as  
80 the failure parameter (Hooputra et al. 2004). Shear damage initiation begins when the shear  
81 damage initiation criterion, which is a function of the equivalent plastic strain  $\epsilon_{eq}$ , is satisfied, as  
82 shown in the following equation

$$83 \quad \int_0^{\epsilon_{eq}^{**}} \frac{d\epsilon_{eq}}{\epsilon_{eq}^{**}(\eta)} = 1 \quad (1)$$

84 Equation (1) is satisfied when the equivalent plastic strain reaches the critical value  $\epsilon_{eq}^{**}$  given by  
85 Equation (2).

$$86 \quad \varepsilon_{eq}^{**} = \frac{\varepsilon_s^+ \sinh [f(\theta_s^+ - \theta_s^-)] + \varepsilon_s^- \sinh [f(\theta_s^+ - \theta_s^-)]}{\sinh [f(\theta_s^+ - \theta_s^-)]} \quad (2)$$

$$87 \quad \theta_s = \frac{(1 - k_s \eta) \sigma_{eq}}{\tau_{max}} \quad (3)$$

88 The parameters  $\varepsilon_s^+$  and  $\varepsilon_s^-$  are the equivalent plastic strains at shear damage initiation for  
89 equibiaxial tensile and compressive deformations, respectively, while  $f$  is an orientation  
90 dependent parameter and  $k_s$  is a material parameter. The first three parameters, which depend on  
91 the material type and the strain rate, “must” be determined experimentally. The variable  $\eta$  is the  
92 stress triaxiality.  $\theta_s^+$  and  $\theta_s^-$  are the shear stress ratios for equibiaxial tension (at  $\eta = \eta^+$ ) and  
93 compression (at  $\eta = \eta^-$ ), respectively, which depend on the maximum shear stress  $\tau_{max}$ , the  
94 equivalent stress  $\sigma_{eq}$  and the stress triaxiality  $\eta$ , as shown in equation 3.

95 Shear damage initiation is associated with the progressive degradation of the elasticity or  
96 material stiffness which commences at the moment of the strain softening of the material at the  
97 point of the ultimate strength/stress (Henning 2012). Shear damage initiation is followed by  
98 damage evolution that is specified in terms of the equivalent plastic displacement,  $U^{pl}$ . Material  
99 failure typified by the complete loss of load-carrying capacity takes place in elements with  
100 completely degraded stiffness, which occurs when the equivalent plastic displacement  $U^{pl}$   
101 reaches the critical value, the equivalent plastic displacement at failure  $U_f^{pl}$ . The equivalent  
102 plastic displacement at failure  $U_f^{pl}$  is specified as zero for instantaneous failure, and greater than  
103 zero (up to unity) for progressive failure. When the material fails, the damage variable  $d$  reaches  
104 its maximum value  $d_{max}$ , and elements with completely degraded stiffness are either retained or

105 removed from the mesh. The default value of  $d_{\max}$  is 1.0, but can be specified by the user as 0.99  
106 (ABAQUS 2014).

### 107 **Calibration of S690 structural steel model parameters using phenomenological curve fitting**

108 Moze et al. (2007) conducted the tension coupon tests on S690 solid and perforated structural  
109 steel coupons, depicted in Figures 1(a) and 1(b), respectively. The hole in the perforated tension  
110 coupon H04 had a diameter of 10 mm, which was the same as the coupon thickness. The  
111 elongation of each coupon was measured by Moze et al. (2007) over a gauge length of 280 mm.

112 The coupon models in Figures 1(c) and 1(d) were meshed with 1mm x 1mm x 1mm C3D8R  
113 element that had been established through mesh convergence studies to be the appropriate  
114 element size. C3D8R elements are 8-node hexahedral linear brick reduced integration elements  
115 with hourglass control. Non-linear quasi-static analyses were conducted using the explicit solver  
116 (ABAQUS 2014) at the default step time of one second to ensure that the kinetic energy is far  
117 less than 5% of the total energy. The true stress-strain curves were calculated from the published  
118 experimental force-displacement (Moze et al. 2007). The elastic modulus is assumed to be 200  
119 GPa, with a Poisson's ratio of 0.3. The inelastic responses of the coupons were modelled with  
120 the isotropic hardening rule and the von Mises yield criterion.

121 The experimental displacement at fracture of the solid coupon was not provided by Moze et al  
122 (2007), so the calibration was conducted on the perforated coupon H04 instead, of which the  
123 experimental displacement at fracture was 6.6 mm. The ultimate test load was 811 kN.

124 The phenomenological curve fitting process was conducted on the perforated coupon model with  
125 the material parameter  $k_s$  and the equivalent plastic displacement at failure  $U_f^{pl}$  being kept



126 constant at 0.1 (Hooputra et al. 2004), while the equivalent plastic strain at shear fracture, shear  
127 stress ratio and strain rate were varied, as shown in Table 1. Parameter combination A was the  
128 experimentally calibrated values employed by Hooputra et al. (2004) for the simulation of ductile  
129 fracture in aluminum alloy EN-AW-7108 TC. Parameter combinations B to E are linear  
130 multiples (2.5, 3.0, 3.5, and 4.0) of the parameter combination A.

131 As shown in Table 2, parameter combination D resulted in the closest estimate of the  
132 experimental displacement at fracture obtained by Moze et al (2007) for the perforated specimen  
133 H04. The experimental load-displacement curve can be compared to that obtained using  
134 parameter combination D in Figure 2.

135 In order to verify that the material parameter  $k_s$  has little if any significance in the S690 steel  
136 tensile responses predicted using the phenomenological fracture model, eleven simulations were  
137 conducted for the specimen H04 using the same parameters as the parameter combination D  
138 except for the material parameter, which was varied from 0 to 10 at a step of 1. It was found that  
139 all eleven simulations returned the same results as the parameter combination D, which had a  
140 material parameter  $k_s$  equal to 0.1.

#### 141 **FE predictions of the ultimate loads and displacements at fracture of perforated coupons**

142 In addition to the solid and perforated coupons H04 discussed in the preceding section, Moze et  
143 al. (2007) tested perforated tension coupons with hole diameters of 5 mm, 13 mm, 18 mm, 22  
144 mm, 26 mm, 30 mm, 40 mm and 50 mm, designated as specimens H03, H06, H07, H08, H10,  
145 H12, H15 and H16, respectively. The load-displacement curves up to the displacement at  
146 fracture predicted using the parameter combination D are shown in Figure 3. The ultimate loads  
147 and displacements at fracture obtained numerically by the authors and experimentally by Moze

148 et al. (2007) are shown in Table 3. The variable  $R_{EP}$  represents the ratio of the experimental  
149 values to the predicted values.

150 Both the experimental tests (Moze et al. 2007) and the present simulations found that the  
151 perforated coupon with a 5-mm hole, H03, exhibited the largest displacements at fracture, being  
152 9.0 mm and 10.7 mm, respectively. Incidentally, the difference between the experimental and the  
153 simulation results is the greatest for this coupon. In any case, the consistently accurate  
154 predictions of the tensile responses of the solid and perforated coupons, as evident in Table 3,  
155 demonstrate that the phenomenological shear damage and fracture model presented in this  
156 technical note is feasible and reliable.

#### 157 **Simulations of ductile fracture under pure shear and under combined shear and tension**

158 In order to establish the ability of the present phenomenological shear damage and fracture  
159 model to predict fractures under pure shear and under combined shear and tension, the pure shear  
160 and the combined shear and tension loading tests of TRIP 690 steel specimens by Bai and  
161 Wierzbicki (2010) were simulated. The fracture simulation were conducted using a true yield  
162 stress of 445 MPa, a true tensile strength of 920MPa, a true yield strain of 0.02 and a true  
163 ultimate strain of 0.26 reported Li et al. (2010). An equivalent fracture strain of 1.1044, a shear  
164 stress ratio of 40, a strain rate of  $0.00040 \text{ s}^{-1}$ , a material parameter of 0.1 and an equivalent  
165 plastic displacement at failure of 0.1 were obtained through the phenomenological curve fitting  
166 process.

167

168 The models of the pure shear specimen and of the combined shear and tension specimen are  
169 shown in Figures 4(a) and 4(b), respectively. The gauge sections of both specimens were meshed

170 with 1mm x0.25mmx 0.25mm elements while the other sections were meshed with 1-mm cube  
171 elements. The loading pins were modeled as 3D analytical rigid cylindrical shells with the  
172 reference points at their base centers. The general contact (explicit) algorithm with the tangential  
173 behaviour penalty frictional interaction and a friction coefficient of 0.15 were employed to model  
174 the contact between the loading pins and the specimens. The top pins were fixed and the bottom  
175 pins were subjected to vertical downward displacements until the specimens fractured.

176 The predicted fracture processes of the specimens are shown in Figures 5 and 6 respectively. The  
177 failure of the specimen subjected to pure shear began with the initiation of a vertical fracture  
178 approximately parallel to the loading direction at the middle of the bottom end of the butterfly  
179 gauge section (Figure 5(a)), where the shear damage initiation criterion designated as SHRCRT  
180 (ABAQUS 2014) first reached its critical value of 1. The vertical fracture propagated upwards  
181 and the model fractured completely as shown in Figure 5(b). The failure of the specimen  
182 subjected to combined shear and tension began with the initiation of a fracture inclined to the  
183 loading direction at the top end of the strip gauge section, as shown in Figure 6(a). The initiated  
184 slanted fracture propagated downwards and the model fractured completely as shown in Figure  
185 6(b).

186 Bao and Wierzbicki, (2004, 2010) were unable to capture the exact location of the fracture  
187 initiation and the fracture propagation. Consequently, the predicted fracture processes of the  
188 specimens subjected to pure shear and to combined shear and tension could not be validated  
189 against their experimental results. However, it can be seen from the good agreements between  
190 the experimental fracture shapes (Bao and Wierzbicki, 2010) and the simulated ones shown in  
191 Figure 7 that the present phenomenological shear damage and fracture model is capable of  
192 modelling fractures under pure shear and under combined shear and tension.

193 **Conclusions**

194 The predictions of structural steel tensile responses and fractures under tension, pure shear and  
195 combined shear and tension using the phenomenological shear fracture model that simulates  
196 ductile shear fracture based on the shear stress ratio (rather than the Lode angle parameter) has  
197 been presented. This technical note has covered the predictions of tensile responses of S690 steel  
198 solid coupons and perforated coupons having different hole diameters. It has also simulated the  
199 fractures of TRIP 690 steel specimens under pure shear and under combined shear and tension.  
200 The calibrated phenomenological shear damage and fracture model parameters, being the  
201 equivalent fracture strain, the shear stress ratio, the strain rate, the equivalent displacement at  
202 fracture and the material parameter were obtained through phenomenological curve fitting  
203 process. It has been found that the material parameter  $k_s$  has no noticeable effect on the shear  
204 damage and fracture, and the tensile responses (ultimate load and displacement at fracture) of  
205 structural steels

206 This technical note has demonstrated that the phenomenological shear fracture model can be  
207 employed in the design office for the predictions of ultimate loads and displacements at fracture  
208 required for the design of resilient or fracture-resistant steel structures, and for simulating the  
209 complete fracture process in structural steel. The phenomenological shear damage and fracture  
210 model parameters can be obtained through phenomenological curve fitting process that does not  
211 require costly laboratory tests.

212

213 Table 1. Parameter combinations for S690 steel

Parameter Combinations	Equivalent Fracture Strain $\epsilon_{eq}^{**}$	Shear Stress Ratio $\theta_s^+$	Strain Rate (s <sup>-1</sup> )	Material Parameter $k_s$	Equivalent Plastic Displacement at Failure
A	0.27610	10	0.00010	0.1	0.1
B	0.69025	25	0.00025	0.1	0.1
C	0.82830	30	0.00030	0.1	0.1
D	0.96635	35	0.00035	0.1	0.1
E	1.10440	40	0.00040	0.1	0.1

214

215 Table 2. Ultimate loads and displacements at fracture of perforated specimen H04

Parameter Combinations	Predicted Load (kN)	Difference (%)	Predicted Displacement at Fracture (mm)	Difference (%)
A	800	1.28	2.97	55.0
B	800	1.28	5.17	21.7
C	800	1.28	5.87	11.1
D	800	1.28	6.59	0.04
E	800	1.28	7.36	11.5

216

217 Table 3. Experimental and FE ultimate loads and displacements at fracture

Specimen	Hole Size (mm)	Ultimate Loads			Displacement at Fracture		
		Test (kN)	FE (kN)	$R_{EP}$	Test (mm)	FE (mm)	$R_{EP}$
H01	0	886	875	1.01	22.5	21.5	1.05
H03	5	861	860	1.00	9.0	10.7	0.84
H04	10	811	799	1.01	6.6	6.5	1.02
H06	13	789	782	1.01	7.0	6.9	1.01
H07	18	747	733	1.02	7.1	6.9	1.03
H08	22	716	702	1.02	7.1	7.3	0.97
H10	26	679	665	1.02	7.9	7.9	1.00
H12	30	641	627	1.02	8.0	7.9	1.01
H15	40	542	542	0.99	8.2	7.8	1.05
H16	50	456	454	1.00	8.4	9.2	0.91

218

219

220 **References:**

- 221 ABAQUS (2014) ABAQUS Analysis User's Manual, Version 6.14, Dassault Systèmes,  
222 Providence RI.
- 223 Bao, Y. and Wierzbicki, T. (2004) "On fracture locus in the equivalent strain and stress  
224 triaxiality space." *Int. J. Mech. Sci.*, 46, 81–98
- 225 Bai, Y. and Wierzbicki, T. (2008) "A new model of metal plasticity and fracture with pressure  
226 and Lode dependence." *Int. J. Plasticity*, 24, 1071–1096.
- 227 Bai, Y., Wierzbicki, T. (2010). "Application of extended Mohr–Coulomb criterion to ductile  
228 fracture." *Int J Fract*, 161:1–20
- 229 Bernauer, G. and Brocks, W. (2002) "Micro-mechanical modelling of ductile damage and  
230 tearing—results of a European numerical round robin". *Fatigue and Frac. of Eng. Mat. &*  
231 *Struct.*, 25(4), 363-384.
- 232 Björklund, O., Larsson, R. and Nilsson, L. (2013) "Failure of high strength steel sheets:  
233 Experiments and modeling." *J. Mat. Proc. Tech.*, 213, 1103–1117.
- 234 Björklund, O. and Nilsson, L. (2014) "Failure characteristics of a dual-phase steel sheet." *J. Mat.*  
235 *Proc. Tech.*, 214, 1190–1204.
- 236 Cheng, L., Monchiet, V., Morin, L., Saxcé, G. and Kondo, D. (2015) "An analytical Lode angle  
237 dependent damage model for ductile porous materials." *Eng. Fract. Mech.*, 149, 119–133.
- 238 Danas, K. and Castañeda, P. P. (2012) "Influence of the Lode parameter and the stress triaxiality  
239 on the failure of elasto-plastic porous materials." *Int. J. Solids and Struct.*, 49, 1325–1342.
- 240 French, I. E. and Weinrich, P. F. (1979) "The shear mode of ductile failure in mild steel." *Mat.*  
241 *Sci. Eng.*, 39 (1), 43-46.



242 Gologanu, M., Leblond, J. B. and Devaux, J. (1993) “Approximate models for ductile metals  
243 containing nonspherical voids – case of axisymmetrical prolate ellipsoidal cavities”. *J. Mech.*  
244 *Phys. Solids*, 41 (11), 1723–1754.

245 Henning, L. (2012). *Simulating Ductile Fracture in Steel using the Finite Element Method:*  
246 *Comparison of Two Models For Describing Local Instability due to Ductile Fracture*, MSc  
247 thesis, Faculty of Mathematics and Natural Sciences, University of Oslo, Norway.

248 Hooputra, H., Gese, H., Dell, H., and Werner, H. A. (2004) “Comprehensive failure model for  
249 crashworthiness simulation of aluminium extrusions.” *Int. J. Crashworthiness*, 9 (5), 449–64.

250 Jia, L. and Kuwamura, H. (2014) “Ductile Fracture Simulation of Structural Steels under  
251 Monotonic Tension.” *J. Struct. Eng.*, DOI: 10.1061/(ASCE)ST.1943-541X.0000944

252 Kanvinde, A. M. and Deierlein, G. G. (2006) “The Void Growth Model and the Stress Modified  
253 Critical Strain Model to Predict Ductile Fracture in Structural Steels.” *J. Struct. Eng.*,  
254 10.1061/(ASCE)0733-9445(2006)132:12(1907).

255 Kiran, R., and Khandelwal, K. ( 2013) “A micromechanical model for ductile fracture prediction  
256 in ASTM A992 steels.” *Eng. Frac. Mech.*, 102, 101-117.

257 Li, W., Liao, F. Zhou, T., and Askes, H. (2016) “Ductile fracture of Q460 steel: Effects of stress  
258 triaxiality and Lode angle.” *J. Construct. Steel Res.*, 123, 1–17.

259 Li, Y., Luo M., Gerlach, J., and Wierzbicki, T. (2010) “Prediction of shear-induced fracture in  
260 sheet metal forming.” *J. Mat. Proc. Tech.*, 210, 1858–1869.

261 Moze, P., Beg, D. and Lopati, J. (2007) “Net cross-section design resistance and local ductility of  
262 elements made of high strength steel.” *J. Construct. Steel Res.*, 63, 1431–1441.

263 Nahshon, K. and Hutchinson, J. W. (2008) “Modification of the Gurson model for shear  
264 fracture.” *Euro. J. Mech. Solids*, 27, 1–17.

- 265 Rice, J. R. and Tracey, D. M. (1969) “On the ductile enlargement of voids in triaxial stress  
266 fields.” *J. Mech. Phys. Solids*, 17, 201-217.
- 267 Wen, H. and Mahmoud, H. (2015) “New Model for Ductile Fracture of Metal Alloys. I:  
268 Monotonic Loading.” *J. Eng. Mech.*, DOI: 10.1061/(ASCE)EM.1943-7889.0001009.
- 269 Wilkins, M. L., Streit, R. D. and Reaugh, J. E. (1980) “Cumulative strain- damage model of  
270 ductile fracture: simulation and prediction of engineering fracture tests.” *Research Report*  
271 *No. UCRL-53058*, Science Applications, San Leandro, CA.
- 272 Xue, L. (2008) “Constitutive modeling of void shearing effect in ductile fracture of porous  
273 materials.” *Eng. Frac. Mech.*, 75, 3343–3366.

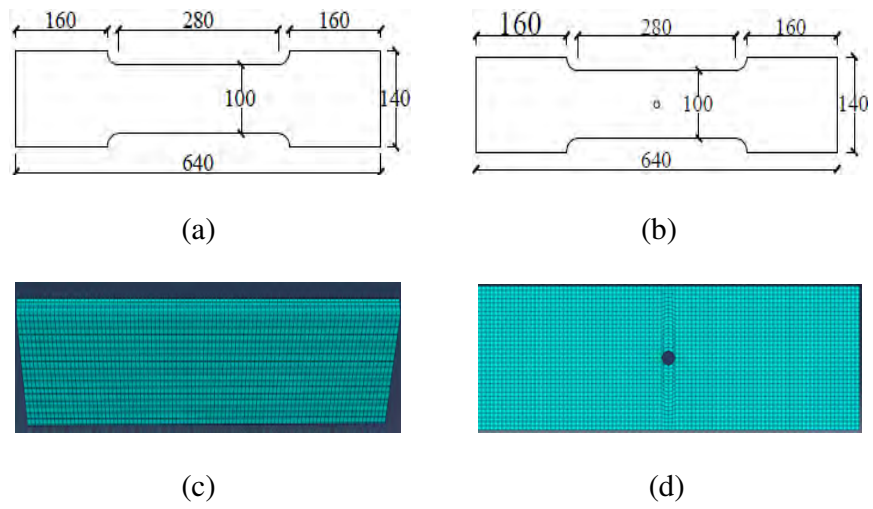


Figure 1 Test specimens and FE models: (a) Solid tension coupon (H01); (b) Perforated coupon (H04); (c) Solid coupon's FE model; (d) Perforated coupon's FE model.

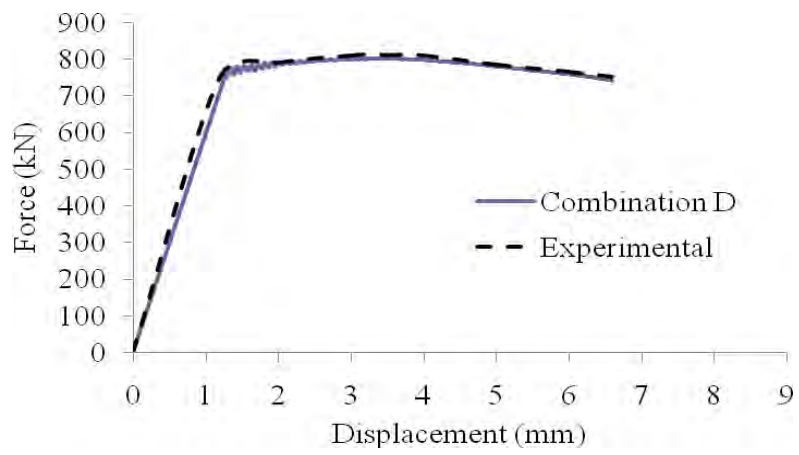


Figure 2 Experimental and analysis results of H04 using parameter combination D

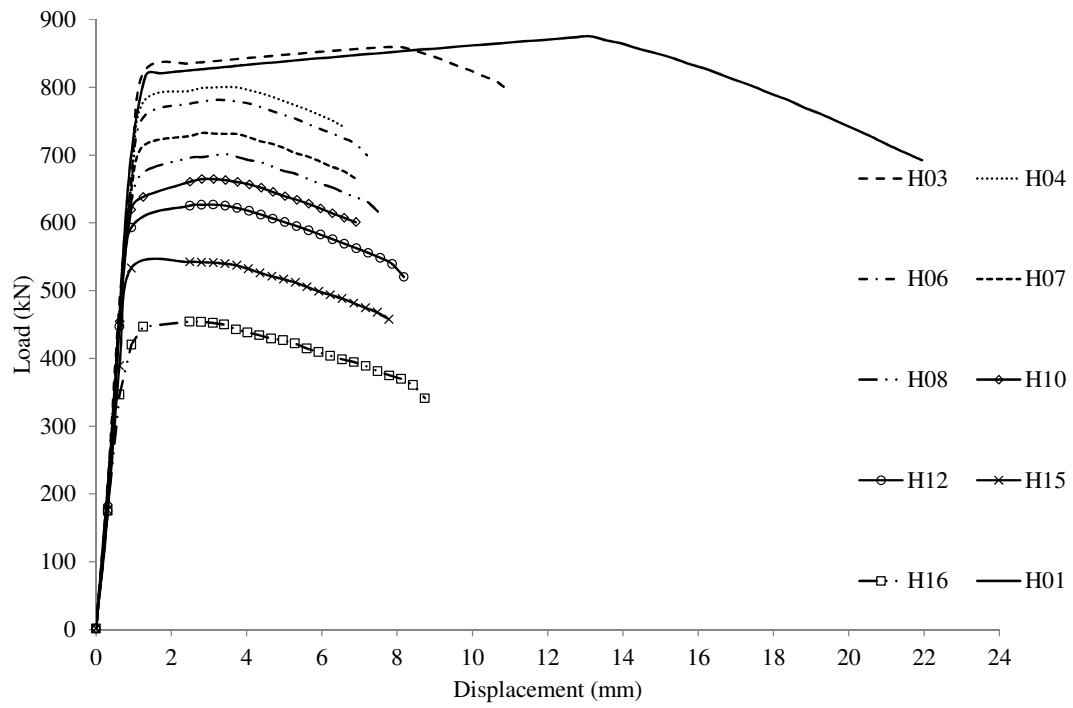


Figure 3 Force-displacement curves for solid coupon (H01) and perforated coupons with varying hole diameters using parameter combination D

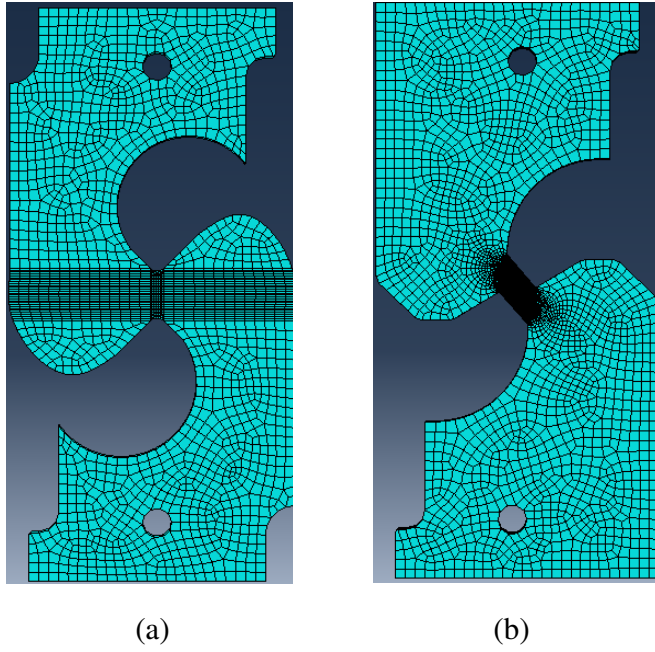


Figure 4: FE models of: (a) Pure shear specimen; (b) Combined shear and tension specimen

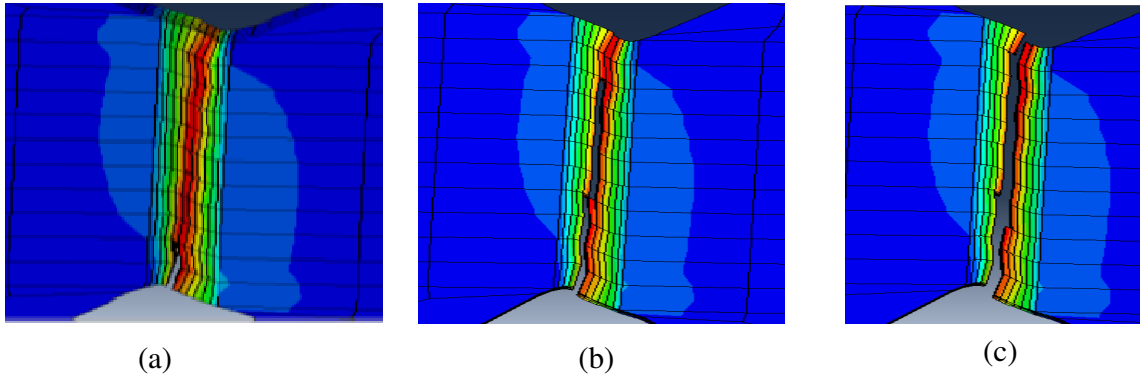


Figure 5: Predicted fracture processes of the specimen subjected to pure shear: (a) Fracture initiation; (b) Fracture propagation; (c) Complete fracture

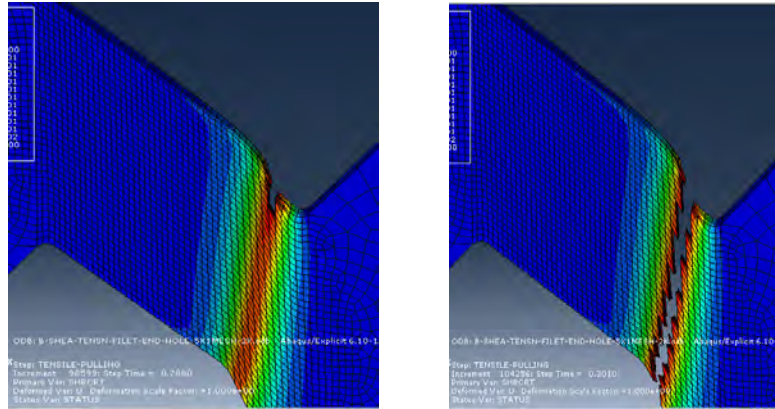


Figure 6: Predicted fracture processes of the specimen subjected to combined shear and tension:

(a) Fracture initiation and propagation; (b) Complete fracture



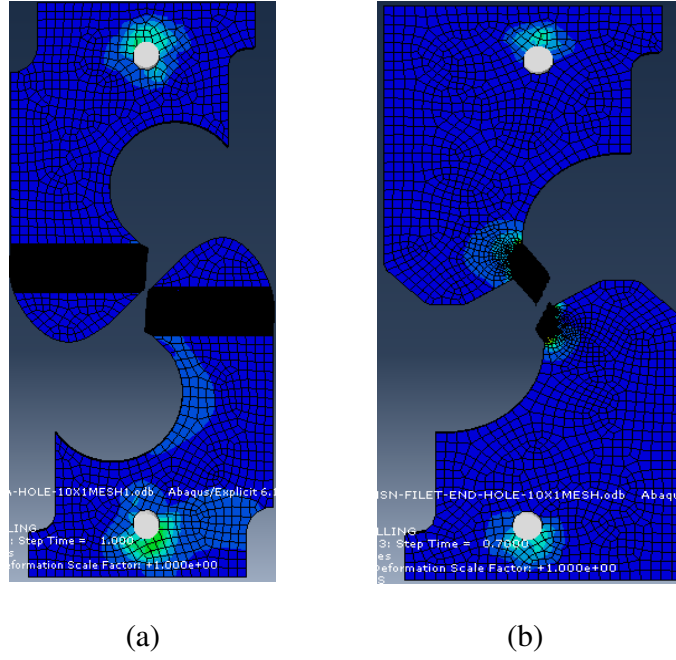


Figure 7: Predicted fractured shapes: (a) Pure shear specimen; (b) Combined shear and tension specimen