

2005

Fabrication and characterization of superconducting PLD MgB₂ thin films

Yue Zhao

University of Wollongong, yue_zhao@uow.edu.au

UNIVERSITY OF WOLLONGONG

COPYRIGHT WARNING

You may print or download ONE copy of this document for the purpose of your own research or study. The University does not authorise you to copy, communicate or otherwise make available electronically to any other person any copyright material contained on this site. You are reminded of the following:

This work is copyright. Apart from any use permitted under the Copyright Act 1968, no part of this work may be reproduced by any process, nor may any other exclusive right be exercised, without the permission of the author.

Copyright owners are entitled to take legal action against persons who infringe their copyright. A reproduction of material that is protected by copyright may be a copyright infringement. A court may impose penalties and award damages in relation to offences and infringements relating to copyright material. Higher penalties may apply, and higher damages may be awarded, for offences and infringements involving the conversion of material into digital or electronic form.

Unless otherwise indicated, the views expressed in this thesis are those of the author and do not necessarily represent the views of the University of Wollongong.

Recommended Citation

Zhao, Yue, Fabrication and characterization of superconducting PLD MgB₂ thin films, PhD thesis, Institute for Superconducting & Electronic Materials, University of Wollongong, 2005. <http://ro.uow.edu.au/theses/816>

Research Online is the open access institutional repository for the University of Wollongong. For further information contact the UOW Library: research-pubs@uow.edu.au

NOTE

This online version of the thesis may have different page formatting and pagination from the paper copy held in the University of Wollongong Library.

UNIVERSITY OF WOLLONGONG

COPYRIGHT WARNING

You may print or download ONE copy of this document for the purpose of your own research or study. The University does not authorise you to copy, communicate or otherwise make available electronically to any other person any copyright material contained on this site. You are reminded of the following:

Copyright owners are entitled to take legal action against persons who infringe their copyright. A reproduction of material that is protected by copyright may be a copyright infringement. A court may impose penalties and award damages in relation to offences and infringements relating to copyright material. Higher penalties may apply, and higher damages may be awarded, for offences and infringements involving the conversion of material into digital or electronic form.

Fabrication and Characterization of Superconducting PLD MgB₂ Thin Films

A thesis submitted in fulfillment of the
requirements for the award of the degree

Doctor of Philosophy

From

University of Wollongong

By

Yue Zhao, M.E.

Institute of Superconducting & Electronic Materials
Faculty of Engineering

2005

Certification

I, Yue Zhao, declare that this thesis, submitted in fulfillment of the requirements for the award of Doctor of Philosophy, in the Institute for Superconducting & Electronic Materials, Faculty of Engineering, University of Wollongong, is wholly my own work unless otherwise referenced or acknowledged. The document has not been submitted for any other academic institution.

Yue Zhao

16 December 2005

Acknowledgement

First of all, I want to thank my supervisors, Prof. Dou and Dr. M. Ionescu, for their instructions and supports through out my PhD study. Dr M. Ionescu also provides major trainings on the operations of PLD deposition system, XRD, and high temperature furnaces.

The Australian Research Council provided the project founding and my APAI scholarship on this work. The University of Wollongong and the Institute for Superconducting & Electronic Material provided financial support to the work.

The magneto-optical imaging in this work was carried out by PhD candidate M. Roussel. The cross-sectional TEM specimen preparation and imaging in Chapter 7 are carried out by Prof. P. Munroe and his colleagues in the Electron Microscopy Unit at UNSW. Here I want to thank their contribution to the thesis work.

Dr. A. V. Pan, Dr. J. Horvat provided major training on MPMS and PPMS low temperature physical properties measurement. Dr. D. Wexler provided TEM training. Dr. K. Konstatinov provided SEM training. Dr. Violeta and Dr. P. Whitton provided AFM training. Here I want to thank their warm-hearted helps, lots of time and great patience.

I want to thank Dr Tania's helps for improving my English scientific article writing skill. She has proof read every single piece of my papers that have been published in various journals.

I have had very useful discussions with Dr. E. W. Collings, Dr. X. L. Wang, Dr. D. Q. Shi, Dr. S. H. Zhou, Dr. Saeid, Dr. Sockat, Dr. G. Alvalas and Dr. T Tajima.

Mr. R. Kennel, Mr. N. Mackie, Mr. G. Tillman and Mr. Stuart have offered strong technical support during my PhD research period.

I would like to thank my parents back in China for their persistent support and encouragements to my study.

Finally, I want to thank my wife, Lu Wei, for her dedicated support and complete understanding throughout my PhD study.

Contents

Certificate of Originality.....	I
Acknowledgment.....	II
Contents.....	III
Abstract.....	VIII
List of Figures.....	XII
List of Tables.....	XXII

Introduction.....	1
Chapter 1. Literature survey.....	6
1.1. Superconductivity in intermediate-temperature superconductor MgB ₂	6
1. 2 MgB ₂ film synthesis methods.....	12
1.2.1 Methods for MgB ₂ film Synthesis.....	12
1.2.1.1 As-grown films.....	12
1.2.1.2 <i>In situ</i> and <i>ex situ</i> annealing methods.....	14
1.2.1.3 Other MgB ₂ film forming methods.....	16
1.2.2 Microstructures in MgB ₂ films from different preparation methods...	17
1.3 Chemistry and physics of MgB ₂ film formation.....	20
1.3.1 Formation of MgB ₂ phase and its stability.....	20
1.3.2 Crystallization of MgB ₂ film.....	21
1.4. Potential applications of MgB ₂ films.....	23
14.1 Electronic devices.....	23

1.4.1.1 Superconducting Josephson Junctions and SQUIDs.....	26
1.4.1.2 Passive Microwave devices.....	27
1.4.2 Coated conductors.....	28
References.....	30
Chapter 2: Experimental Methods.....	40
2.1 MgB ₂ thin film preparation.....	40
2.1.1 Pulsed laser deposition system for MgB ₂ thin film preparation.....	40
2.1.2 <i>In situ</i> annealing procedure.....	45
2.1.3 <i>Ex situ</i> annealing procedure.....	45
2.2 Microstructure detection.....	46
2.2.1 Scanning Electron Microscopy detection.....	46
2.2.2 Transmission Electron Microscopy detection.....	48
2.2.3 X-ray diffraction analysis.....	52
2.2.4 Atomic force microscopy observation.....	53
2.3 Magnetic and transport properties measurements.....	55
2.3.1 The PPMS and the MPMS systems.....	55
2.3.2 Four-probe transport measurement.....	67
2.3.3 Magneto-optic Imaging analysis.....	68
References.....	62

Chapter 3 PLD MgB₂ thin films using <i>in situ</i> annealing.....	64
3.1 The function of argon background gas.....	65
3.2 The influence of annealing temperature and heating rate on the T _c	70
3.3 The laser energy and the target and substrate distance.....	74
3.4 The change of target surface morphology after laser ablation.....	76
3.5 The J _c and flux penetration behaviour of the on-axis MgB ₂ films.....	81
3.6 Microstructure of the <i>in situ</i> MgB ₂ films.....	83
3.7 Transport measurement results.....	91
3.7.1 Resistivity dependence on temperature.....	91
3.7.2 Irreversibility field, upper critical field, and flux flow activation energy.....	93
3.8 Discussion on the depressed T _c in the <i>in situ</i> MgB ₂ films.....	96
3.9 Summary.....	100
References.....	101
Chapter 4 Comparative study on <i>in situ</i> and <i>ex situ</i> annealed MgB₂ thin films.....	106
4.1 Experimental.....	106
4.2 Microstructural difference between the <i>ex situ</i> and <i>in situ</i> MgB ₂ films.....	108
4.3 Transport properties of the two types of films.....	112
4.3.1 Superconducting transition temperatures and residual resistivity.....	112
4.3.2 Field dependence of ρ(T) curves for the two types of MgB ₂ films.....	114
4.4 Superconducting behaviors in magnetic fields for the <i>in situ</i> and <i>ex situ</i> films..	120
4.4.1 Dependence of critical current density on applied field and temperature.....	120
4.4.2 Flux penetration behavior detected by magneto-optical imaging.....	121

4.5 Discussions on the differences of H_{c2} , J_c and magnetic behavior between the two types of film.....	122
4.6 Summary	127
References.....	128
Chapter 5: Modification of superconducting properties of the MgB_2 film by Si addition.....	132
5.1 Introduction.....	132
5.2 Si addition in the MgB_2 films using PLD deposition technique.....	134
5.3 Si content and distribution in the Si-added MgB_2 films	135
5.4 T_c and J_c dependence on the level of Si addition	138
5.5 Irreversibility field, upper critical field and residual resistivity in the Si-added films	142
5.6 Discussion on the correlation between the performance enhancement and the Si addition level.....	144
5.7 Conclusions.....	145
References.....	146
Chapter 6. Off-axis MgB_2 films using an <i>in situ</i> annealing pulsed laser deposition method.....	150
6.1 Fundamentals of off-axis deposition.....	151
6.2 Experimental details of the off-axis PLD deposition in this work.....	153
6.3 Optimization of the off-axis preparation parameters.....	155
6.3.1 The influence of laser energy and growth rate on the surface morphology of off-axis MgB_2 film.....	155

6.3.2 The influence substrate temperature and annealing conditions on the T_c of the off-axis films.....	157
6.4 Microstructural properties of deposited off-axis films.....	160
6.4.1 Atomic Force Microscopy results on surface topography of the off-axis films.....	160
6.4.2 The texture of the off-axis MgB_2 film.....	161
6.4.3 Transmission Electron Microscopy and Selected Area Electron Diffraction results of the off-axis film.....	163
6.5 Resistivity, H_{irr} , H_{c2} , activation energy and the critical current density of the off- axis MgB_2 films	163
6.6 MOI results of the off-axis MgB_2 films in comparison with on-axis films.....	169
6.7 Summary.....	173
References.....	174
Chapter 7 Multilayer MgB_2/Mg_2Si thin film.....	179
7.1 MgB_2/Mg_2Si multilayer preparation by using an off-axis target-switching PLD deposition method.....	180
7.2 Structural characterization on the multilayer film.....	181
7.3 Transport properties of the Multilayer film.....	183
7.4 Arrhenius plot and the activation energy of flux flow for the multilayer film...	186
7.5 Summary.....	188
References.....	190
8. Summary and conclusions.....	193
List of Publications during PhD study period.....	196

Abstract

The aim of this thesis was to study the thin film of magnesium diboride (MgB_2) superconductor based on PLD synthesis technique. The MgB_2 is a very simple binary compound with a number of surprising properties. The discovery of superconductivity with a critical temperature (T_c) of 39 K in bulk MgB_2 in January 2001 has attracted a huge research interests worldwide in this material. In a hope to substitute MgB_2 superconducting electronics for low temperature superconducting electronics and compete with high temperature superconductors, the preparation techniques of MgB_2 thin films need to be advanced from a material engineering point of view. On the other hand, new studies regarding the different behaviours of MgB_2 thin films and MgB_2 bulks will assist the scientific community to better understand the physics in this superconductor.

We began the study with *in situ* MgB_2 film preparation using normal on-axis geometry. The *in situ* annealing conditions of pulsed laser deposited MgB_2 films were studied. We found that the superconducting properties depend in a crucial way on the annealing conditions: temperature, heating rate and time. We tested the T_c dependence of the *in situ* annealed MgB_2 films by changing various process parameters, including laser energy density, target-substrate distance, background gas, annealing temperatures, heating rates and dwell times. The film processing conditions were optimized and good quality *in situ* films were obtained routinely, with good reproducibility. The hysteresis loops of magnetic moment versus applied field at different temperatures indicate a weak field dependence in high fields. Magneto-optical imaging of the films showed quite homogeneous magnetic flux penetration, indicating structural homogeneity. The films without annealing show no superconductivity.

Another method using *ex situ* annealing has also been tested for a better crystallization of the MgB₂ film, as well as for reference purposes. In the *ex situ* annealing process, we first deposited boron precursor film on a sapphire substrate and then wrapped it in Ta foil and sealed it in a stainless steel tube together with pure Mg pellets, under protective Ar atmosphere. The tube was then annealed in tube furnace and kept at 900°C for 30 min. With this method, we obtained *ex-situ* MgB₂ films of comparable quality as those reported so far in the literature, and used them as a benchmark to test possible improvements in MgB₂ thin film technology based on PLD.

Significant differences in properties between the *in situ* films and *ex situ* films were found. The *ex situ* annealed MgB₂ film has a T_{c onset} of 38.1K, while the *in situ* film has a suppressed T_{c onset} of 34.5K. The resistivity at 40K for the *in situ* film is larger than that of the *ex situ* film by a factor of 6. The residual resistivity ratios are 1.1 and 2.1 for the *in situ* and *ex situ* films respectively. The field dependence of the resistivity-temperature curves has been measured. A large slope of the H_{c2}-T curve was obtained for the *in situ* annealed film. The J_c-H curves of the *in situ* film show a much weaker field dependence than those of the *ex situ* film, attributable to stronger flux pinning in the *in situ* film. The microstructural differences between the two types of films are observed by AFM and TEM. The small-grain (<60nm) size and a high oxygen level detected in the *in situ* annealed MgB₂ films may be decisive for the significant improvement of J_c and H_{c2}.

In order to enhance the performance of the MgB₂ films, various amounts of Si up to a level of 18wt% were added into MgB₂ thin films fabricated by pulsed laser deposition. Si was introduced into the PLD MgB₂ films by sequential ablation of a stoichiometric MgB₂ target and a Si target. The T_c's of the Si added MgB₂ thin films

were tested. A J_c enhancement was observed in the Si added MgB_2 films. For the ~3.5wt% Si addition, the best enhancement circumstance, the magnetic critical current density (J_c) of the film at 5K was increased by 50% as compared to the undoped film. The slope of $H_{irr}(T)$ and $H_{c2}(T)$ curves of the 3.5wt% Si added MgB_2 film was slightly higher than that for the undoped film.

For the application in superconducting electronics, the surface smoothness of the MgB_2 thin film is of crucial importance. We pioneered an off-axis deposition geometry in the PLD MgB_2 films preparation. Highly smooth and c -axis oriented superconducting MgB_2 thin films were successfully achieved with a off-axis geometry. The films were deposited on Al_2O_3 -C substrates, aligned perpendicular to a stoichiometric MgB_2 target in a 120 mTorr high purity Ar background gas. An *in situ* annealing was carried out at 650°C for 1 min in a 760 Torr Ar atmosphere. Despite the short annealing time, an x-ray θ -2 θ scan shows fairly good crystallization, according to the clear c -axis oriented peaks for the films. Both atomic force microscopy and the x-ray diffraction results indicated that the crystallite size is less than 50nm. The root mean square roughness of our off-axis film was ~4 nm in a $5 \times 5 \mu m^2$ area. The zero resistance T_c value of the best off-axis film reached 32.2 K with a narrow transition width of 0.9 K. The films showed no anisotropy in H_{c2} - T curves when parallel and perpendicular fields were applied relative to the film surface. The slope of H_{c2} - T curves is ~1 T/K, which is still among the highest reported values.

On the basis of successful preparation of smooth off-axis MgB_2 films, we obtained MgB_2/Mg_2Si multilayer structure by sequentially switching a stoichiometric MgB_2 target and a Si target during off-axis pulsed-laser deposition. The transmission electron microscope cross-sectional image of the resulting film exhibits a layered structure with each MgB_2 layer being 40-50 nm thick and the Mg_2Si inter-layers about

5 nm thick. A clear enhanced anisotropy in the irreversibility lines and the vortex activation energy was observed. Pinning and the flux flow activation energy for this type of film was significantly increased in parallel applied fields.

List of Figures

Fig.1-1 MgB₂ crystal structure.

Fig.1-2 Experimental specific heat data (○) as a function of the reduced temperature t ($=T/T_c$) from two groups [17, 29]. The data are compared with the BCS-normalized specific heat (thin curve) and also two-gap fits (thick curve). Insets: gaps $2\Delta/k_B T_c$ and $2\Delta S/k_B T_c$ versus t (dotted curves) and partial specific heat of both bands (full curves).

Fig.1-3 Fermi surface of MgB₂ in reciprocal space. The two cylindrical sheets are Fermi surfaces of σ -band.

Fig.1-4 Calculated phase diagram of Mg and B.

Fig.1-5 Morphology change due to relative temperature of metallic film made by evaporation.

Fig.2-1 The schematics of the PLD setup used for this work. The right figure is a 3-D illustration of the on-axis deposition geometry for MgB₂ in situ annealed film. The target-substrate distance is adjustable by changing the height of the heater supporting frame before mounting the part to the chamber.

Fig.2-2 XRD spectrum of the stoichiometric MgB₂ target. The vertical lines indicate the peak positions for MgB₂.

Fig.2-3 The homemade seal-in-argon apparatus for the ex situ annealing of MgB₂ film.

Fig.2-4 (left) JEOL JEM 2010 TEM at UoW; (right) Philips CM200 TEM at UNSW.

Fig.2-5 The steps for planar TEM specimen preparation in this work.

Fig. 2-6 The FIB system (FEI xP200) at UNSW and an illustration of the steps used for FIB preparation of a cross-sectional TEM specimen. a) Sample definition and rough milling, b) Fine milling and final polish, c) Cut, d) Lift out.

Fig.2-7 Concept of AFM and the optical lever: (left) Beam deflection system, using a laser and photodetector to measure the beam position; (right) The SEM of a standard tip.

Fig.2-8 The Dimension 3100 AFM (Digital Instruments) at UoW.

Fig. 2-9 PPMS (left) and MPMS (right) from Quantum Design.

Fig.2-10 Illustrations for four-probe setup for transport measurement in PPMS. The points labeled "I" were current contacts, and the points labeled "V" were voltage contacts. (Left) Magnetic field is perpendicular to the film surface circumstance. (Right) Magnetic field is parallel to the film surface. In both cases the testing current flowing in the MgB₂ film is perpendicular to the applied field.

Fig.2-11 (left) An illustration of Faraday effect and (right) a typical MOI set up.

Fig.2-12 The MOI system in ISEM.

Fig.3-1 The plumes from a stoichiometric MgB₂ target in a 120mTorr argon background gas. On the left the base vacuum was 9×10^{-8} Torr, and on the right 7×10^{-7} Torr. The laser energy was 350mJ/pulse.

Fig.3-2 Laser plumes at different Ar pressures. a) 1×10^{-6} Torr; b) 80mTorr; c) 100mTorr; d) 120mTorr; e) 200mTorr; f) 260mTorr; g) 290 mTorr. The laser fluence was 350mJ/Pulse.

Fig.3-3 The laser plume of MgB₂ on-axis deposition. The argon pressure was 120mTor, the laser fluence was 350mJ and the target-substrate distance was 23mm.

Fig.3-4 The MS spectrum of Ar atmosphere for the chamber at the total pressure of 1×10^{-3} Torr. Two argon peaks are present at 40 and 20 amu. The H₂O peak is at 18 amu, and the H₂ peak is at 2 amu.

Fig. 3-5 T_c dependence on in situ annealing temperatures. The T_c was measured by DC susceptibility in ZFC. The thickness of the thicker films is $\sim 1 \mu\text{m}$; The thickness of the thinner films is $\sim 0.5 \mu\text{m}$. The heating rate of both groups of films was $\sim 110^\circ\text{C}/\text{min}$ (rapid heating) with a 1 min dwell time, followed by a free cooling at $50^\circ\text{C}/\text{min}$ (the power is switched off).

Fig. 3-6 T_c values of MgB₂ films versus in situ annealing ramp time (bottom axis) and heating rate (top axis). The annealing temperature is $\sim 680^\circ\text{C}$ with 1min dwell time.

Fig. 3-7 The DC susceptibility (ZFC-FC) curves for three different target-substrate distances. Sample #260303 was prepared with 500 mJ/pulse laser energy and 40mm target-substrate distance; the #280503 was prepared with 400 mJ/pulse laser energy and 30mm target-substrate distance; and the #190903 was prepared with 300 mJ/pulse laser energy and 20mm target-substrate distance. All three films have a similar thickness of about 300-400 nm.

Fig. 3-8 SEM images of laser ablated MgB₂ target surface. a) A general look at the ablated ring under 50x magnification. b) Central part of the ablated ring ; c) The margins of the ablated ring . d) Area adjacent to the ablated ring .

Fig.3-9 Area-scan EDS spectrum of the MgB₂ target before and after 100 pulses $5 \text{ J}/\text{cm}^2$ 248 nm laser ablation. a) before ablation, b) after ablation, c) on the base of the cones, d) on top of a cone. The inset figures are correlative SEM images of the EDS scanning area.

Fig.3-10 SEM cross-section image of the in situ MgB₂ film.

Fig.3-11 AFM 3D mode image of the MgB₂ film surface. Scale= $500 \times 500 \text{ nm}^2$

Fig.3-12 TEM bright field image of a planar specimen for the in situ MgB₂ film (a). b) is a SAD pattern from a circular 500 nm area of the film.

Fig.3-13 Magneto-optical image of the in situ MgB₂ film at B_a=8.7mT at 20K after zero field cooling. The film size is 3x3mm². The white round spots are defects in the MO indicator. The bright area in the upper middle part of the film is an enhanced flux penetration due to an accidental mechanical scratch on the MgB₂ film.

Fig.3-14 MOI images of in situ annealed film 030703, showing the flux penetration as a function of temperature: (a) 4 K, (b) 7 K, (c) 10K. The applied field is about 10mT for the three sets of observations. The sample is ~3x3 mm² in dimensions.

Fig.3-15 (a) Hysteresis loops of DC magnetisation for our best on-axis in situ film 190903 at 5K(■), 10K(●), 15K(▲), and 20K(▼) respectively and (b) the calculated J_c of the in situ film from the magnetic hysteresis loops at 5K(■), 10K(●), 15K(▲), and 20K(▼) respectively. (c) detailed magnetization curves at 5 K within the field range of 1000 Oe (0.1 T) with very fine scan of 5 Oe resolution. The inset is a magnification of the curve between 400 Oe and 500 Oe.

Fig.3-16 The resistivity versus temperature curve for a typical on-axis in situ MgB₂ film. The thickness of the film is 450 nm.

Fig.3-17 Field dependence of resistivity-temperature curves. The applied field is (from right to left) 0T, 0.1T, 0.5T, 1T, 2T, 3T, 4T, 5T, 6T, 7T, 8T, and 8.7T respectively.

Fig.3-18 Irreversibility field and upper critical field of the in situ MgB₂ film.

Fig.3-19 Activation energy of the in situ MgB₂ film. The grey lines are U₀ data of bulk-MgB₂ cited from ref [1].

Fig.3-20 EDS result of a typical in situ MgB_2 film with 30K magnetic T_c . In order to avoid the interaction with the Al_2O_3 substrate, a low electron beam energy of 5kV was used. The Al $K\alpha$ signal at 1.5 keV is not noticeable in the spectrum.

Fig.4-1 SEM cross-section images of the two types of films. a: in situ annealed MgB_2 film; b: ex situ annealed MgB_2 film on Al_2O_3 -R substrate.

Fig.4-2 AFM 3D images of a) the boron precursor film, the lower plane is the substrate; b) the ex situ annealed MgB_2 film on Al_2O_3 -R substrate; c) the ex situ film on Al_2O_3 -C substrate; d) the ex situ film on 4H-SiC(0001) substrate.

Fig.4-3 XRD of ex situ annealed films on a) Al_2O_3 -R and b) Al_2O_3 -C substrates. The MgO signal may come from the oxidization of some excess metal Mg on the film surface.

Fig.4-4 (a): Temperature dependence of resistivity for the two types of MgB_2 films from 5K to 300K in zero field; The inset shows the transition curves between 30K and 40K. (b): The magnetization versus temperature curves. ■: in situ film; ○: ex situ film.

Fig.4-5 Field dependence of $\rho(T)$ curves for a) in situ MgB_2 film and b) ex situ film.

Fig.4-6 Arrhenius plots of the resistance $R(H, T)$ of the a) in situ and b) ex situ films. The applied field is perpendicular to the film plane.

Fig.4-7 The flux flow activation energy U_0 versus applied field B_a . Two stages of field dependence of U_0 is found in ex situ film.

Fig.4-8 Irreversibility lines and upper critical field versus temperature curves for the in situ and ex situ annealed films. A): Irreversibility lines for the in situ and ex situ films. The data for undoped ex situ MgB_2 film and oxygen alloyed MgB_2 film (Ref. [2]) is displayed in the figure for comparison. B): Upper critical fields versus temperature for the two films. The data for c-axis-oriented MgB_2 films from Ref. [3] are also shown in this figure.

Fig.4-9 Critical current of the two types of MgB_2 films calculated from M-H loops.

Solid symbols: in situ annealed film; lines: ex situ annealed film. The temperature is 5 K, 10 K, 15 K, and 20 K from top to bottom, respectively. The applied field is perpendicular to the film plane.

Fig.4-10 MO images for the in situ film, shown in (a), (b) and the ex situ film, shown in (c), (d) at 4 K. The applied field is (a) 17 mT, 1st set of measurement, (b) 17 mT, 2nd set of measurement, (c) 25.5 mT, 1st set of measurement, and (d) 25.5 mT, 2nd set of measurement.

Fig.4-11 AFM of a) the in situ and b) the ex-situ MgB₂ film on Al₂O₃-Rsubstrate.

Fig.4-12 EDS analysis results for the a) in situ and b) ex situ films. In order to avoid the interaction with the Al₂O₃ substrate, a low electron beam energy of 5kV was used. As a result, no Al signal at about 1.5 KV is detectable. Since the thickness of both films is similar, the difference in oxygen signal intensity can reveal the difference of oxygen level in the two films.

Fig.4-13 AFM 3D image of the ex situ MgB₂ film. The surface topography shows typical randomly oriented grains. The arrow shows a thinner part formed between two grains.

Fig.5-1 AFM deflection image of Si islands on sapphire-R substrate deposited for 50 pulses (10Hz, 5sec). The laser fluence is 300mJ/pulse.

Fig.5-2 EDS Si mapping in the Si doped MgB₂ films. The upper part contains SEM secondary electron images, and the lower part the distribution of Si. a: 3.5% Si addition, b: 11% Si addition. The arrows indicate the Si-rich spots.

Fig.5-3 DC magnetization curves of the films with different Si doping levels. The applied field is 25 Oe for both ZFC and FC measurements.

Fig.5-4 J_c values with different Si doping levels. a: at 5 K, b: at 10 K, c: at 15 K. The applied field B_a is perpendicular to the film plane.

Fig.5-5 Irreversibility lines and upper critical fields of the 3.5% Si doped and undoped MgB₂ films.

Fig.5-6 The resistivity versus temperature curves in fields from 0T to 8.7T. a: undoped film; b: 3.5wt% Si film; c: 5wt% Si film.

Fig.5-7 The J_c and ρ_{40K} versus Si addition level in the doped films. The solid squares represent the magnetic J_c at 5K and 1.5T. The open circles represent residual resistivity ρ_{40K} .

Fig.6-1 Illustration of on axis deposition and off-axis deposition.

Fig.6-2 Kennedy 's design of the off-axis deposition of smooth YBCO film.

Fig.6-3 Schematic Illustration of the off-axis deposition geometry for MgB₂ thin film deposition.

Fig.6-4 SEM cross-sectional images of four films on Al₂O₃ –C substrates using different deposition conditions, namely (a): on-axis deposition, laser energy flux = 300 mJ/pulse, laser repetition frequency = 10 Hz, growth rate=12 Å/sec; (b) off-axis, E=500mJ/pulse, F=10Hz, 16Å/sec; (c) off-axis, E=300mJ/pulse, F=10Hz, 4 Å/sec; and (d) shaded off-axis, E=500mJ/pulse, F=5Hz, 2 Å/sec. The scale is the same for all four images.

Fig.6-5 The ZFC magnetization curves for the off-axis MgB₂ films prepared with different annealing temperatures. The substrate temperature during deposition is 250°C for all the samples. The ramp time from 250°C to the annealing temperature is 12 min and the dwell time is 1min.

Fig.6-6 ZFC magnetization curves of off-axis MgB₂ films deposited on different substrate temperatures. The in situ annealing condition is the same as above-mentioned optimized parameters.

Fig.6-7 $T_{c \text{ onset}}$ versus $T_{\text{substrate}}$ curve (left axis) and Magnetization (right axis) versus $T_{\text{substrate}}$ at 5K. The T_c and Magnetization values are extracted from ZFC magnetization curves at 5 K in Fig.6-6.

Fig.6-8 AFM deflection image of a $2 \times 2 \mu\text{m}^2$ area of the surface of film (d) in Fig.6-4.

Fig.6-9 AFM cross-section profile of the off-axis film #300604V. The vertical distance between the highest and the lowest part indicated by arrows is 22nm.

Fig.6-10 XRD θ - 2θ pattern of the off-axis deposited film #300604V with a slow scanning rate of 0.2 degree/min. The vertical lines label the positions for all MgB_2 peaks in powder diffraction database. The unknown peak at 37.56° is also presents in the spectrum of a bare Al_2O_3 -C substrate, so we assume it is not from the film.

Fig.6-11 The left figure: Bright field TEM image of a planar specimen a typical off-axis MgB_2 film with a zero-resistivity T_c of 32K. The right figure is a SAD pattern from a $\Phi 500$ nm area of the film

Fig.6-12 Resistivity versus temperature for an off-axis film #300604V. The inset is an enlargement of the transition part.

Fig.6-13 The field dependence of resistivity-temperature curves of the off-axis MgB_2 film #300604V in a) perpendicular ($H//c$ axis) fields and b) parallel fields ($H//a$ - b plane).

Fig.6-14 The Arrhenius plot of resistance $R(T, H)$ for the off-axis MgB_2 film in a) perpendicular and b) parallel fields.

Fig.6-15 The activation energy U_0 of flux flow versus applied field, B_a for film #300604V.

Fig.6-16 The H_{c2} -T curves for $H // ab$ -plane and $H \perp ab$ -plane. The H_{c2} values are derived from transport curves using 90% ρ_{Tc} values. In both $H // ab$ -plane and $H \perp ab$ -plane cases, the testing current was perpendicular to the applied field.

Fig.6-17 Magnetic J_c versus applied field for the off-axis film #300604V at different temperatures. The applied field is perpendicular to the film plane. It is difficult to estimate J_c at 5 T in low fields owing to the predominant magneto-thermal instability.

Fig.6-18 MOI of an typical off-axis MgB_2 film with a zero-resistivity T_c of 32K.

Fig.6-19 MOI of the on-axis MgB_2 film #030703 (a) 4K 3.4 mT (b) 4K 8.5 mT (c) 4K 25.5 mT; (d) 7.5K, 5.1mT (e) 7.5K, 10.2 mT (f)7.5K, 17 mT; (g)15K, 3.4 mT (h)15K, 10.2mT (i)15 K, 27.2 mT.

Fig.6-20 MOI image details of a) on-axis film 7.5K 10.5mT, and b) off-axis film at 7K 14.2 mT.

Fig.7-1 Cross-sectional SEM image of the multilayer film. The inset is a BF TEM image of the multilayer film.

Fig.7-2 TEM BF image of the multilayer film. The scale bar is 100 nm. The inset is an SAED of the MgB_2 film, showing a clear textured grain orientation.

Fig.7-3 Resistivity versus temperature curves of multilayer film and MgB_2 film.

Fig.7-4 Resistivity versus temperature curves of a multilayer film in: a) perpendicular fields; b) in parallel fields.

Fig.7-5 The H_{c2} versus T/T_c for the multilayered film and MgB_2 film.

Fig.7-6 The irreversibility fields of the multilayer film and the MgB_2 film.

Fig.7-7 The Arrhenius plot of resistance $R(T, H)$ for the multilayer film in: a) perpendicular and b) parallel fields.

Fig.7-8 The activation energy, U_0 of flux flow versus applied field, B_a .

List of tables:

Table 3-1 T_c values of MgB_2 films versus laser energy and target-substrate distance, D_{T-S} .

Table 5-1 Si content and corresponding T_c in the in situ annealed MgB_2 films produced by PLD.

List of Publications during PhD study period (from 2002 to 2005)

1. **Y. Zhao**, S. X. Dou M. Ionescu P. Munroe, “Significant improvement of activation energy in the MgB₂/Mg₂Si multilayer film”, accepted by Appl. Phys. Lett., To be published in Feb. 2006
2. **Y. Zhao**, S. X. Dou M. Ionescu P. Munroe, “Transport properties of multilayered MgB₂/Mg₂Si superconducting thin film” accepted by J. Appl. Phys. To be published in Jun 2006
3. **Y. Zhao**, M. Ionescu, M. Roussel, a. V. Pan, J. Horvat, and S. X. Dou, “Superconducting and microstructural properties of two types of MgB₂ films prepared by pulsed laser deposition, IEEE Transactions on Applied Superconductivity” **15**, 3261 (2005).
4. **Y. Zhao**, M. Ionescu, J. Horvat and S. X. Dou, “Off-axis MgB₂ films using an *in situ* annealing pulsed laser deposition method”, Supercond. Sci. & Technol. **18**, 395-399 (2005)
5. S. X. Dou, S Soltanian, **Y. Zhao**, E Getin, Z Chen, O. Shcherbakova and J. Horvat “The effect of nanoscale Fe doping on the superconducting properties of MgB₂” Supercond. Sci. Technol. **18**, 710 (2005)
6. M. Roussel, A. V. Pan, A. V. Bobyl, **Y. Zhao**, S. X. Dou, and T. H. Johansen, Magnetic flux penetration in MgB₂ thin films produced by pulsed laser deposition, Supercond. Sci. Technol. **18**, 1391 (2005).
7. **Y. Zhao**, M. Ionescu, J. Horvat and S.X. Dou “Comparative study of *in situ* and *ex situ* MgB₂ films prepared by pulsed laser deposition” Supercond. Science & Technology **17**, S482-S485 (2004)

8. **Y. Zhao**, M. Ionescu, J. Horvat, A.H. Li and S.X. Dou “ Si addition in *in situ* annealed MgB₂ thin films by pulsed laser deposition” Superconductor Science & Technology **17** 1247-1252 (2004)
9. M. Ionescu, **Y. Zhao**, M. Roussel, S.X. dou, R. Ramer and M. Tomsic “Flux pinning in MgB₂ thin films grown by pulsed laser deposition” Journal of Optoelectronics and Advanced Materials, **6**, 169-176 (2004)
10. A.V. Pan, **Y. Zhao**, M. Ionescu, S.X. Dou, V.A. Komashko, V.S. Flis, and V.M. Pan, "Thermally activated depinning of individual vortices in YBa₂Cu₃O₇ superconducting films", *Physica C*, **407**, 10 (2004)
11. M. Ionescu, A. H. Li, **Y. Zhao**, H. K. Liu, A. Crisan, "Enhancement of critical current density in YBa₂Cu₃O_{7-d} thin films grown by PLD on YSZ (001) surface modified with Ag nano-dots", J. Phys. D: Applied Physics, **37**, 1824 (2004)
12. **Y. Zhao**, M. Ionescu, A.V. Pan, S.X. Dou and E.W. Collings “*In situ* annealing of superconducting MgB₂ films prepared by pulsed laser deposition” Supercond. Sci. Technol. **16**, 1487-1492 (2003)