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W.M. Chen
Nanjing University, China

G. Li
University of Wollongong

S. S. Jiang
Nanjing University, China

H. Y. Ling
University of Wollongong

P. Yao
University of Wollongong, peiyao@uow.edu.au

See next page for additional authors

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Transmission electron microscopy evidence for phase transformation from Bi$_2$Sr$_2$CuO$_6$ to Bi$_2$Sr$_2$Ca$_2$Cu$_3$O$_{10}$

W. M. Chen$^{a)}$

Department of Physics and National Laboratory of Solid State Microstructures, Nanjing University, Nanjing 210093, People’s Republic of China and Institute for Superconducting and Electronic Materials, University of Wollongong, NSW 2522, Australia

G. Li

Institute for Superconducting and Electronic Materials, University of Wollongong, NSW 2522, Australia

S. S. Jiang

Department of Physics and National Laboratory of Solid State Microstructures, Nanjing University, Nanjing 210093, People’s Republic of China

H. Y. Ling, P. Yao, M. J. Qin,$^{b)}$ H. K. Liu, and S. X. Dou

Institute for Superconducting and Electronic Materials, University of Wollongong, NSW 2522, Australia

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The growth mechanism of the Bi-2223 phase was studied using multifilamentary Ag/Bi-2223 tapes manufactured by the powder-in-tube method. X-ray powder diffraction was performed to analyze the developments of Bi-2201, Bi-2212, and Bi-2223 phases during sintering periods at high temperatures. Transmission electron microscopy (TEM) was used to investigate the growth mechanism of the 2223 phase. TEM images indicated that 2201 and Ca$_2$CuO$_3$ crystals come together to form platelets of 2223. The structural phase transformation established a structural frame of the 2223 crystal. There were some vacant defects retaining in the established 2223 crystal cells at copper and oxygen sites after the phase transformation, and the Cu/O compound at high temperatures would diffuse into 2223 cells and occupy the corresponding vacancies to complete the 2223 crystal structure. The present work provides an evidence for growing the 2223 phase from the 2201 phase and Ca$_2$CuO$_3$ in the samples studied. © 2002 American Institute of Physics.

There are three compounds in the Bi system: Bi-2201, Bi-2212, and Bi-2223, in which 2223 is of the greatest interest due to its high $T_c$ and the potential for commercial applications. The understanding of the formation mechanism was important for the efficient preparation of Bi-2223 superconductors with high quality. It has been found that the 2223 phase may grow from the Bi-2212 phase by solid state reaction at high temperatures.$^{1–13}$ The primary mechanism of growing the Bi-2223 phase was thought to be the Ca/CuO$_2$ layers intercalating into Bi-2212 crystals at high temperatures.$^{2–4}$ A corresponding layer rigidity model of growing the 2223 phase was suggested.$^{5,6}$ Some different mechanisms of growing the Bi-2223 phase were also suggested, such as the growth of the 2223 phase directly from liquid amorphous$^4$ or in the form of 2212→2201→2223.$^{14}$ Up to now, people only widely accepted that the 2223 phase might grow from the 2212 phase. Whereas, it was believed that the processes of growing 2223 were not thoroughly revealed, because the phase diagrams describing the growth of polycrystalline 2223 were complicated.$^{13–17}$

It was believed that if some of the 2223 phase did not grow from the 2212 phase, such that the 2223 phase might retain some trace of its origins. The trace would leave a record giving some information on the mechanism, which could be observed in transmission electron microscopy (TEM) image fields.

The starting material was a commercial oxide powder, with a stoichiometry of Bi(Pb):Sr:Ca:Cu=2:2:2:3. The starting powder was thoroughly ground and became amorphous materials.$^{18}$ The Bi-2223 compound was prepared in the silver sheathed tapes, because the 2223 phase in the multifilament tapes has a better preferential orientation than in bulk,$^{19}$ and a good orientation of grains has advantages for TEM observation.

Multifilamentary Ag/2223 tapes were manufactured by the powder-in-tube method.$^{20–22}$ The sintering temperature was 840 °C, for 2, 10, 20, 30, and 50 h.

Figure 1 shows patterns of x-ray diffraction for tapes with sintering periods of 2, 10, 20, 30, and 50 h, respectively. Figure 1 shows records of the 2201, 2212, and 2223 phase forming, developing, and/or transforming, during the sintering period at 840 °C. It is found that the development of the 2212 phase is faster than that of the 2201 and 2223 phases. 2212 is the main phase during the first 10 h sintering and then transforms rapidly as the sintering period increases, so that at the end of 50 h sintering it only constitutes a trace amount in the tape. The development of the 2201 phase is similar to, but more slowly than that of 2212. It reaches its maximum amount at the end of 30 h sintering and then be-
gins to disappear, so that it is less than 5% in amount after 50 h sintering. The 2223 phase growth proceeds throughout the sintering period. It appears when 10 h sintering and both 2223 and 2212 are the main phases after 20 h sintering. 2223 by the end of 30 h sintering becomes the main phase and is completed at the end of 50 h sintering.

Since 2201, 2212, and 2223 phases have well preferential orientation on c axis, the strength of the characterized diffraction peaks [006], [008], and [0010] may quantitatively describe the development of 2201, 2212, and 2223 phase, respectively. The relative changes of peaks [006], [008], and [0010] during the sintering period are listed in Table I. Percent strengths of the corresponding peaks are relative to that of the maximum peak in each pattern and the maximum is determined to be 100%. X-ray analysis shows that dissolving the 2201 and/or 2212 phases both finally contribute to the 2223 phase, the two second phases on the left of the image have been determined to be 2201 and Ca$_2$CuO$_3$. The 2223 grain is growing from left to right. There are second phases remaining on the left. There are transition regions visible between the 2223 phase and the two second phases.

When searching the TEM field of view, the 2201, 2212, and 2223 phases as well as some impurity phases, such as Ca$_2$CuO$_3$ and CuO, were often observed. A special morphology of platelets consisted of the 2223 and 2201 phases and a Ca$_2$CuO$_3$ compound was found and shown in Fig. 2. It is noted that there is no 2212 phase around the grown 2223 platelets, but 2201 and Ca$_2$CuO$_3$ phases are found close to the 2223. Physical intuition tells us that the 2223 phase may come from the combination of the 2201 phase with Ca$_2$CuO$_3$.

TEM further focused on the combined zone of the three phases and a more detailed image was taken and shown in Fig. 3. 2223, Ca$_2$CuO$_3$, and 2201 platelets are found in both Figs. 2 and 3. In Figs. 2 and 3, the bright dot lines sloping from the upper left to the lower right are the crystal boundary between Ca$_2$CuO$_3$ and 2201. We find a special transition region with high brightness at the upper left in Fig. 3, in which Ca$_2$CuO$_3$ and 2201 platelets at the region of their boundary are coming together and become a new crystal lattice in the form of 2223 platelets. The transition region in the form of triangularity extends from the upper right to the lower middle part of Fig. 3. Platelets in the transition region are developing into matured 2223 crystals and the 2223 crystal in Fig. 3 is growing from left to right. Therefore, Fig. 3 shows the evidence that the 2223 phase can develop from the 2201 phase combining with Ca$_2$CuO$_3$ compound.

Similarly, the process of insertion Ca–Cu–O Bi-layers into the existing 2212 to form 2223 can be accomplished by the diffusion through the edge at the growth fronts of the layers. An intercalation mechanism was also invoked for the phase conversion of Bi-2201 to Bi-2212 from an amorphous precursor. It is noted the similarity existing among the conversions of 2201 and 2212 into 2223 and that of Bi-2201 to Bi-2212. For the highly anisotropic crystals, such as Bi-2212 and Bi-2223 crystals, it appears reasonable that platelet insertion mechanism is the mechanism for the phase conversion.

The growth 2223 phase through the combination of the Ca$_2$CuO$_3$ compound

![Image 1](https://via.placeholder.com/150)

**FIG. 1.** The 2212 phase develops rapidly before 20 h sintering, then is quickly eliminated, almost disappearing by the end of 50 h sintering. Similarly to 2212, 2201 develops and reaches its maximum amount at the end of 30 h sintering. It is then eliminated, and only a trace amount remains after 50 h sintering. The 2223 phase grows steadily throughout sintering and has almost replaced the 2201 and 2212 phases at the end of 50 h sintering.

**FIG. 2.** There are three phases in the grain. The main portion of the grain is the 2223 phase, the two second phases on the left of the image have been determined to be 2201 and Ca$_2$CuO$_3$. The 2223 grain is growing from left to right. There are second phases remaining on the left. There are transition regions visible between the 2223 phase and the two second phases.

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**TABLE I. Development of phases depends on the sintering period.** For tapes with good preferential orientation, the composition of each phase in the tape may be characterized by the corresponding characteristic peak. Values of corresponding characteristic peaks are relative to that of the maximum peak in each pattern and the maximum is determined to be 100%.

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2201 phase with Ca$_2$CuO$_3$ compound constitutes an entirely different mechanism. It is noted that the lattice constants for the c axis of the 2201 phase and the Ca$_2$CuO$_3$ crystal are, respectively, 2.51 and 1.22 nm, and the sum of the two is 3.73 nm, the same dimension as the c-axis lattice parameter of the 2223 crystal cell. Obviously, the growth mechanism of the 2223 phase is a structural phase transformation, and, in this case, it does not cause a change of dimensions and is known as the second type of phase transformation. The second type of phase transformation does not require extra mechanical energy to drive the transformation. It means that the phase transformation may be completed without any pressure. Thus, the transformation can arise under the circumstances of the sintering processes that take place during the manufacture of Bi-2223 tapes. Furthermore, this phase transformation is completed through the route of physical diffusion at high temperatures.

In view of the chemical reactions, the reaction should be in the form of

$$\text{Bi}_2\text{Sr}_2\text{Cu}_2\text{O}_8 + \text{Ca}_2\text{CuO}_3 + \text{CuO} = \text{Bi}_2\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10} - \text{CuO}. \quad (1)$$

Whereas, there is no evidence of CuO to be found in the observed microregion. It is noted that this problem we face involves a physical structure transformation, not a pure chemical reaction. It means that this transition establishes a crystal structure with the space group 14/mmm of the 2223 crystal, and its ideal stoichiometry shall be Bi$_2$Sr$_2$Ca$_2$Cu$_3$O$_{10}$. Since there is no CuO to be found in the reaction region, the frame of the crystal structure has been established after the transformation occurred. This implies that there should be two vacancies; one in a copper site and the other in an oxygen site, existing in each 2223 crystal cell immediately after the transformation is completed. The established crystal frame has vacant defects with it and, therefore, it is not perfect. If we discuss this problem from the view of the chemical reaction as well, Eq. (1) should be changed into the form of

$$\text{Bi}_2\text{Sr}_2\text{Cu}_2\text{O}_8 + \text{Ca}_2\text{CuO}_3 + \text{CuO} = \text{Bi}_2\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10} - \text{CuO}. \quad (2)$$

After 30 h sintering, the 2223 crystal with the vacant defects of copper and oxygen sites becomes the main phase, and the required CuO exists in the tape, because of the starting material for producing the 2223 compound having a stoichiometry of the ratio of Bi(Pb):Sr:Ca:Cu = 2:2:2:3. It is noted that the dimensions of the CuO crystal cell are only 0.34–0.52 nm, much less than that of Bi$_2$Sr$_2$Cu$_2$O$_6$ or Ca$_2$CuO$_3$. At high temperatures, smaller cells diffuse easily. Therefore, it is believed that the required copper and oxygen atoms will diffuse into 2223 cells and occupy the corresponding vacancies to complete the 2223 crystal structure after long period sintering at high temperatures.

In summary, TEM images have provided evidence for the phase transformation from 2201 and Ca$_2$CuO$_3$ crystals to Bi-2223, through the diffusion of Ca$_2$CuO$_3$ platelet pieces into 2201 platelets at high temperatures in the samples studied. Each 2223-crystal cell established after the phase transformation has two vacant defects, one in a copper site and the other in an oxygen site. At high temperatures, the required copper and oxygen atoms will diffuse into 2223 cells and occupy the corresponding vacancies to complete the 2223 crystal structure.